

GEAR TECHNOLOGY



MARCH/APRIL 2000

The Journal of Gear Manufacturing

HEAT TREATING ISSUE

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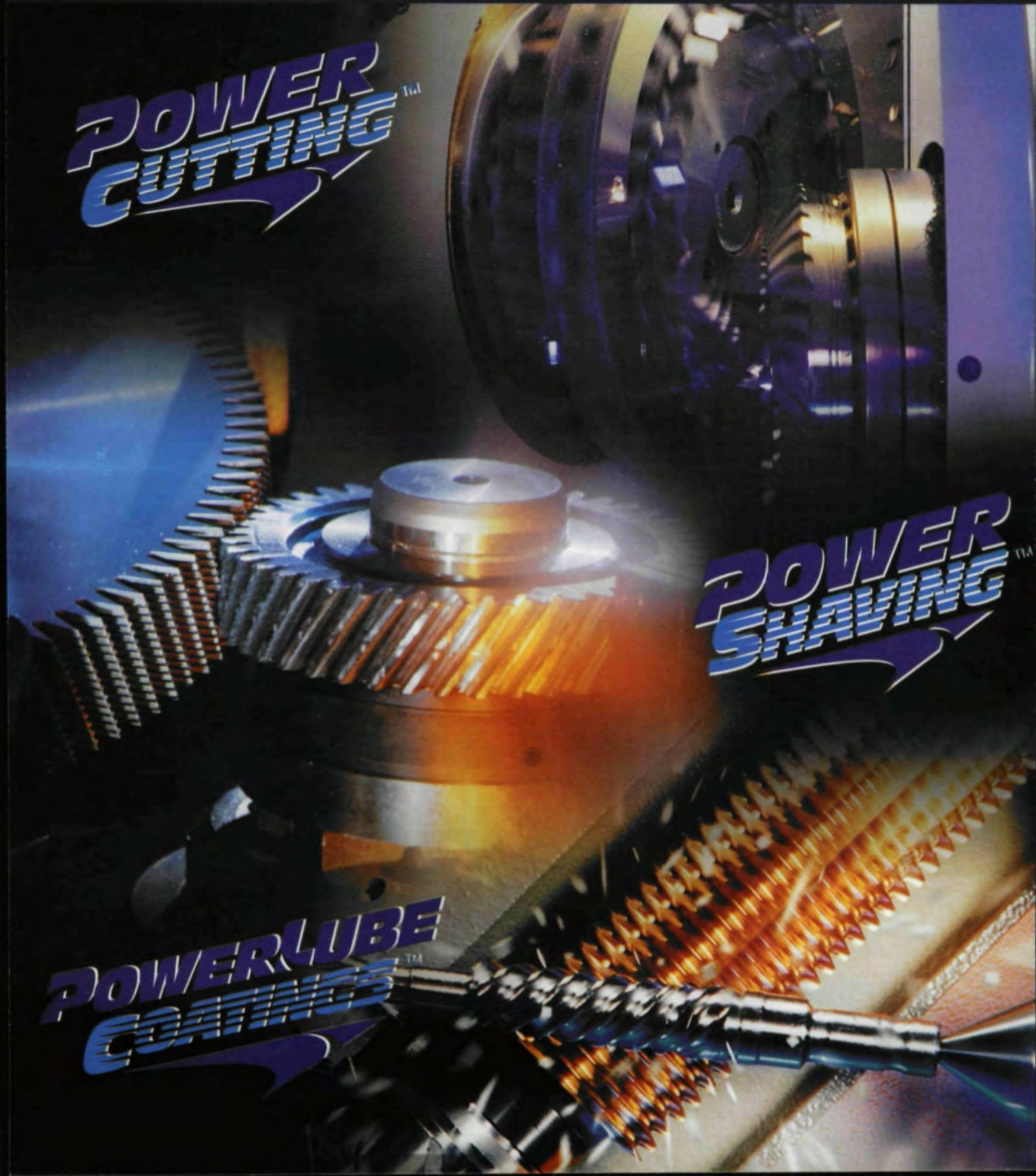
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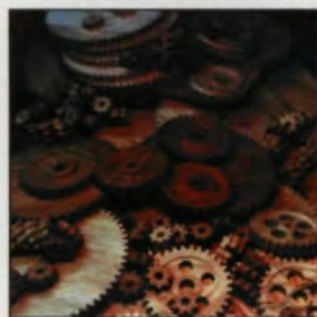
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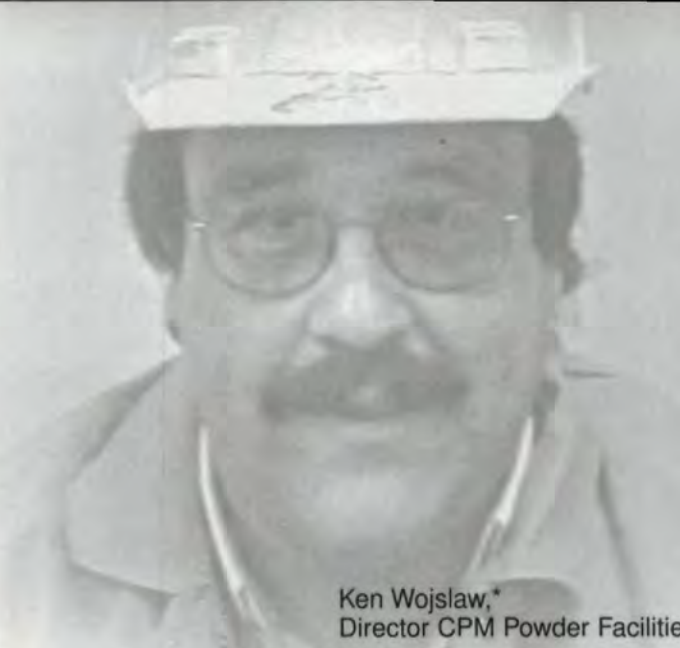
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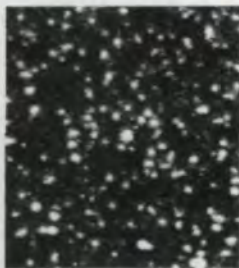
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Goldstein's Paradox

—AND WHAT WE'RE DOING ABOUT IT

I just got off the phone with an associate of mine at a large gear manufacturing company. I was congratulating him on being awarded a new contract when he told me that they had just experienced a substantial downsizing.

As we went through the list of *Gear Technology* subscribers at his plant, I had to remove almost half the names, including a fair number of high-level engineering and management titles, because they were no longer there.

I shouldn't be surprised at the layoff, though. The shrinking of the manufacturing sector has been well publicized. For example, according to a recent *Business Week* article, manufacturing industries cut an average of 26,000 workers per month through the first 10 months of 1999.

Despite this shrinkage of the workforce, manufacturing continues to grow in terms of sales volume and actual output. This is as true in the gear industry as in other manufacturing sectors. In its September/October newsletter, the American Gear Manufacturers Association published numbers from the 1997 U.S. Economic Census. These numbers indicate that open gear manufacturers employed just over 16,000 workers in 1997, compared to more than 25,000 in 1977. However, over the same time period, the value of annual product shipments rose from \$2 billion to \$2.4 billion (in 1997 dollars).

One set of numbers makes it look like the gear industry is doomed to fade away, while the other demonstrates its continued health. That's Goldstein's Paradox, Part I.

The explanation is really quite simple: We've gotten much more productive. I liken the situation to that of agriculture in our country's early history. With today's modern farming equipment, one worker produces far more wheat, corn or rutabagas than his predecessor of a century ago. What used to take the manual labor of many—ploughing, digging, planting and harvesting—is now accomplished by a few using equipment and technology.

Now we're seeing the same thing in our own industry. It wasn't so long ago that every gear machine had an operator standing by it. Today, because of increased automation, better tools and CNC controls, a single operator can run several machines. In addition, today's rigid machines and carbide tools are capable of cutting much faster than earlier models. We've become more efficient off the shop floor as well. Years of investment in computers, software and processes for CAD/CAM, just-in-time inventory control, order processing and business management have resulted in huge gains in productivity, and the Internet will likely provide even greater productivity in the years to come. The result is that we can produce far more gears with far fewer people.

However, the demand for skilled gear industry workers is still very high. You'd think that with layoffs like those my associate on the phone described, there should be a lot of skilled gear people available for hire. But one of the biggest complaints I hear from gear companies is that finding skilled people is becoming harder and harder. That's Goldstein's Paradox, Part II.

One explanation for this paradox is that changes in technology make it a constant job just to keep up. Gear manufacturing companies are forced to continually train and retrain their employees. Education is important, but *continuing* education is crucial.

A couple of weeks ago, I visited the AGMA Training School for Gear Manufacturing, where I was not surprised to learn that demand for gear training is as high as ever. The classes are held at the campus of Richard J. Daley College in Chicago, where instructors teach machinists, engineers and others the basics of gear manufacturing in a hands-on machine shop setting. Other gear-related semi-

nars and classes on our technical calendar each issue seem to indicate that these types of programs are going strong as well.

Although the changing technology explains part of the paradox, another factor also plays a role. Although it seems logical that layoffs in the gear industry and a continual contraction of the workforce should provide for extra workers, unemployment in our country is at an all-time low. Most of those gear people are being absorbed by other industries.

It seems to me that our industry should do all it can to keep the skilled employees it already has. How does someone who's been downsized find a new job that takes advantage of his skills? How does a gear company know where to find the plant managers, gear engineers, operators, quality assurance people, sales engineers and others it needs? If we had some central place where the talent and the companies looking for it could communicate, we might be able to solve some of that problem.

That's why I've decided to announce something new being offered for free through *The Gear Industry Home Page™*. We've created an online service wherein qualified gear industry people who are out of work can post their qualifications and contact information at www.geartechnology.com at no cost. Managers at gear companies can locate the skilled workers they need, and employees who become the victims of downsizing or corporate restructuring can find employment that takes full advantage of the skills they've acquired.

This service is only for gear industry employees who have been laid off or have been given notice. A job seeker who wants to take advantage of our site must post the name of his previous employer and supervisor. We're not trying to create a situation where employees begin querying their companies' competitors for positions. We're trying to help recycle talent back into the industry.

This service can also be a benefit to the companies that are forced to cut staff. Very often, employees who get laid off are excellent, productive workers, who under other circumstances would still have their jobs. It's our hope that the managers or human resource departments of these companies will be able to help these workers find new jobs in the gear industry by making them aware of our free service and helping them post their information online.

Improving communications between employers and potential candidates will help solve part of the problem. By embracing the tools of the information age, we can help the gear industry build on the strength it already has.



Michael Goldstein

Michael Goldstein, Publisher and Editor-in-Chief



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CIRCLE 112

Heat Treating on a Grand Scale

The gear, a double helical with a 106" outside diameter and a 36" face, weighs 40,000 pounds. Destined for use as an intermediate drive gear for a hot strip mill in Latin America, the specifications called for an effective case depth of 0.250" and a nominal surface hardness of HRC 58-62 with a surface carbon content of 0.70-0.90%. The manufacturer turned to Metlab of Philadelphia, PA, to do the job.

According to James Conybear, one of the owners of Metlab, "We get orders from all over the world and heat treat two or three huge gears a month." His partner, Mark Podob, adds, "We carburize and nitride gears for mining equipment, mills, cement plants, ship drives, and the military. There is only one other furnace in the world that can accommodate the sizes we work with."

When working with such large workpieces, Podob and Conybear say that the real key is uniformity of heating. Without it, the process is imprecise and unrepeatable. Three major areas of concern determine the uniformity of the heating process: fixturing, quenching and process control. If these areas are not carefully watched, there can be problems. "With these big assemblies," says Conybear, "their weight will crush them. The material tends to sink into itself,

almost self-forging, with the inside material moving toward the outside." That leads to distortion. "Most gear designers can design for dimensional change," says Podob, "so the key is to make the process precise and the results repeatable." That is accomplished by proper fixturing, quenching and process control.

Fixturing. "You have to make sure the part stays straight," says Podob. "With a big gear, you can pick it up with a three-point loading if, in the furnace, it is resting on a flat plate." With proper support, the slumping and tooth distortion effects of self-forging can be kept to a minimum. "With splines," adds Podob, "the easiest way is to support the tooth section and control the heating to minimize distortion."

Process Control. This covers the precise, uniform heating and cooling (quenching) cycle, as well as the atmosphere within the furnace itself. "This is very important because these gears are in the carburizing furnace for 8 to 10 days," says Conybear. "We have to know what is happening to them so we can make adjustments as the cycle continues." This is accomplished using samples of the same material as the workpiece as well as a standard sample that Metlab uses to gauge how well the process is working. According to Podob, throughout the cycle, these material samples are cut and inspected. Diffusion calculations are then

Welcome to Revolutions, the column that brings you the latest, most up-to-date and easy-to-read information about the people and technology of the gear industry. Revolutions welcomes your submissions. Please send them to Gear Technology, P.O. Box 1426, Elk Grove Village, IL 60009, fax (847) 437-6618 or e-mail people@geartechnology.com. If you'd like more information about any of the articles that appear, please circle the appropriate number on the Reader Service Card.

performed to check the progress of the carburization process. In the case of the 40,000 pound gear mentioned above, the workpiece and samples were of 4320 steel and they underwent 200 hours (8.3 days) of carburization.

Precise heat soak and quench control are important to minimizing tooth distortion. "The key is to minimize the differential temperatures within the part," says Podob. "If you heat big gears too fast, the inside moves to the outside." You also have to cool them down just as carefully, quickly moving the hot workpiece to the quench tank before it can begin to cool. "You need to have a large volume of quenching fluid, usually oil, with good circulation and temperature control," says Podob. "It also helps to have good material to begin with."

Not all materials are suitable for this kind of application, so making the right choice of material takes on added importance when designing very large gears. "We are often involved in the design stage," says Podob. "When you design one of these large components, you have to be more of a materials engineer to make sure that the workpiece will make it through the heat treating process with a minimum of distortion."

Circle 250



40,000 lb. double helical gear on its way to the quenching tank. Courtesy of Metlab.

The SPocket Variable-Ratio Transmission

Say the words "continuously variable transmission" and most people familiar with the concept will envision V-belts and a split pulley. The ratio is changed by increasing or decreasing the split between the two pulley halves. This works because the distance of the V-belt from the axis of rotation and the transmission ratio are proportional to the separation of the pulley halves.

Continuously Variable Transmissions Today. The problem with a belt-driven system is low torque capability, since torque depends on the frictional force that maintains the V-belt's contact with the pulley. This is solved by using corrugated belts and pulleys or chains and sprockets to provide positive, non-slip engagements. For such an arrangement to be truly continuously variable, a complex kind of corrugated pulley or

sprocket with a continuously variable diameter and a constant rib pitch would be needed. While the advantages of using continuously variable transmission technology are very real, the drawbacks of present designs—limited torque, expense and complexity, size and weight, are problematic. What's more, according to Vince Bakulich, president of Revolution Industries of Santa Monica, California, many present designs are not really continuously variable at all. They adjust from one ratio to the next incrementally. Also, many designs are simply not practical or adaptable to everyday use. Recognizing the need for a simple, continuously variable transmission suitable for high-torque applications, Bakulich thinks he has come up with a better way.

The SPocket. Bakulich has designed the SPocket, a continuously variable gear or power transmission that operates at any discrete ratio within a finite range while constantly engaged and under load. Referred to as either an Infinitely Variable Transmission or a Continuously Variable Ratio Transmission (CVRT), the SPocket's corrugated belt maintains engagement while the gear circumference expands and contracts. "There is no slippage," said Bakulich. "You have a direct connection between the chain or belt and the driving member. Take a motorcycle. You have a chain and a sprocket—a solid connection." Bakulich then explained that in a motorcycle you have an internal transmission with three or four ratios. If, however, you could change the size of the sprocket, you could have an infinite number of ratios within a given range. Its ability to change the circumference of the gear is what permits variable speeds and power transmission. This change in circumference, called actuating the gear, can be accomplished in a number of ways, according to Bakulich, including hydraulically, mechanically, electrically or pneumatically. "It really depends on the application and the size and weight requirements." Then he added that the applications for this technology are very real and practical.

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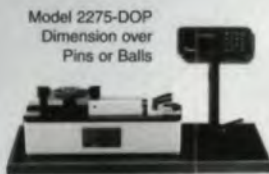
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"Imagine getting in your car," said Bakulich, "and pressing the accelerator to get the engine up to 1,500 RPM, or whatever you need to provide enough power, and leaving it there at a constant speed. Then, you press a button up to go faster or down to go slower." Bakulich explained that with a conventional automobile, to go faster you have to accelerate the engine. In an automobile equipped with a CVRT, you would be adjusting the transmission instead of the engine speed.

CVRT Advantages. This example points out one of the main benefits of continuously variable transmissions—fuel efficiency and energy conservation. Other features of the SProcket design include precise gear ratio selection, continuous power transmission over the entire range of gear ratios with no belt slipping, high versatility and strong, simple construction. The prototype was built with readily available, off-the-shelf parts. Also, the speed at which gear ratios change can be altered allows the CVRT to go through its entire range of gear ratios as slowly or as quickly as necessary.

Applications and Response. According to Bakulich, the greatest use of conventional continuously variable transmissions is in the auto industry. Other applications include utility and constant velocity motors and HVAC units, bicycle chains and sprockets, motorcycle and electric vehicle transmissions, pumps, and wide variety of others. "The SProcket can be used on a wide variety of applications," said Bakulich, "but a unit would need to be designed to fit each one individually. We have had some interest from industrial and automotive manufacturers, but not too much response yet from Detroit."

Circle 251

Pushing the Envelope With Plastic

Achieving higher power density in an existing design space is one of the chief goals of today's gear designer. Because of the economic and design advantages offered by plastic materials, plastic gear

engineers are developing some of the most innovative solutions.

Kleiss Gears, Inc., of Shoreview, MN, has developed some designs that promise increased torque capacity, reduction ratio and strength, says company president Roderick Kleiss.

Kleiss has worked with gear design-consultant Alex Kapelevich to develop a general purpose design for asymmetrical plastic gears. Although the designs

haven't been used in production yet, they were developed as a possible solution for a lawn sprinkler manufacturer who had problems with broken gear teeth because of a rugged product assembly process.

Kleiss Gears actually worked on two designs to increase the strength of the gear teeth, the asymmetrical set and a beefed-up symmetrical set with increased pressure angle. According to Kleiss, both sets performed well in their

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CIRCLE 169

tests. "The asymmetrical set was obviously stronger than the symmetrical set," Kleiss says. However, the manufacturer chose to go with the standard gears because of the disadvantage of having to ensure the asymmetrical gears were going the right direction during assembly.

Despite this, Kleiss believes the asymmetrical design shows great promise for future applications. "We expect to see a 15-30% increase in strength," Kleiss says.

Kleiss Gears has also developed an orbiting, or planocentric, gear design that promises to deliver high torque output and high reduction in a relatively small package. The company is exploring the possibility of using this design in several applications, including a trolling motor steering mechanism, Kleiss says.

According to Kleiss, this application is especially well suited for plastic. "Internal gears always face limitations with cut gearing," Kleiss says. With plastic, the designer can develop the teeth for maximum strength without having to worry about the tooth generating process. The high pressure angle in this set would be extremely unusual in cut metal gears, Kleiss says.

The orbiting design is still in the development phase. Kleiss is working on ways to overcome the uneven torque distribution on the carrier pins, and the company is developing an output coupler that will help to balance the bearing load.

Another issue that often concerns plastic gear designers is the heat generated by the drive system. "Heat is the



An asymmetrical gear set. Courtesy of Kleiss Gears.



Planocentric gear set. Courtesy of Kleiss Gears.

killer," Kleiss says. Although the gear teeth in the orbiting gear set do not generate much heat, they have to be insulated from the heat generated by the motor, Kleiss says.

Kleiss sees much promise for this design. "I think this is just the beginning," he says. These gearsets will be ideal for developing a compound differential, Kleiss says, because they'll offer an extremely high reduction in a small package with good radial load output. ⚙

Circle 252

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Metlab Acquires Potero

Jim Conybear and Mark Podob, owners of Metlab, a commercial heat treating company with an emphasis on large components, deep case carburizing and nitriding, have purchased the assets and ongoing business of John V. Potero, Inc., specialists in atmosphere heat treating small to medium sized batches, induction hardening and black oxide processing. According to Podob, "The combination offered a true win-win business opportunity by strengthening the value offered to both customer bases. Metlab customers benefit from black oxide treatments and a well established pick-up and delivery service. Potero customers gain increased capacity to assure fast turnaround as well as the ability to handle larger parts and deeper case depths."

Gleason-Pfauter Italia S.p.A. Sale Completed

European Kinetic Systems (E.K.S.) B.V., a Dutch subsidiary company of Paritel S.p.A., an Italian corporation, has purchased Gleason-Pfauter Italia S.p.A. The terms of the agreement have not been disclosed. E.K.S. will change the name to DE.CI.MA. S.p.A and the newly acquired company will serve as a contract manufacturer to the parent for certain of the products currently produced at Gleason-Pfauter Italia's facilities.

Keough Named to Board of ASM Heat Treating Society



John R. (Chip) Keough, owner and CEO of Applied Process, Inc., has been named to the ASM Heat Treating Society (HTS)

board of directors. President and owner of Applied Process since 1992, Keough holds seven heat treating or foundry-related patents. He was recognized as an ASM Fellow in 1998 and was awarded the Wm. J. Grede Award by the American Foundrymen's Society that same year.

Two other HTS board members were also named: Daniel H. Herring, director of research and development at Ipsen International, and Ronald A. Wallace,

chief technologist-modeling at Wyman Gordon Co. All of the new board members will serve three-year terms.

Cleveland Gear and PIV Sign Partnership Agreement

Cleveland Gear Company, a subsidiary of the Vesper Corporation, recently signed an agreement with PIV Antrieb Werner Reimers GmbH, of Bad Hombur, Germany. The agreement authorizes Cleveland Gear to manufacture and market the PIV Posired II series of enclosed drives to North American

customers. Cleveland Gear will market these drives as their Redi II product line. The agreement includes Cleveland in PIV's global partnership network and enables Cleveland to provide complete drive systems employing PIV-designed drives in combination with a variety of mechanical power transmission components. "This partnership allows us to fully integrate Redi II drives into complete systems that are performance matched to meet customer requirements," said John Atkinson, Cleveland's vice president of Sales and Marketing.

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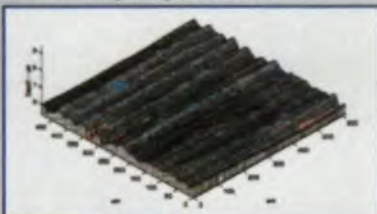


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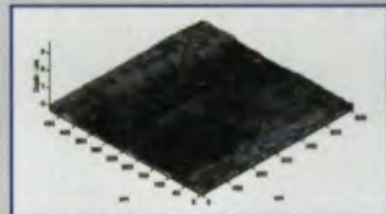
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New Directors at Inductoheat

Inductoheat, Inc., a Madison Heights, Michigan, manufacturer of induction heating equipment and systems, has announced the following promotions:



Ed Haddad

Former project engineer Ed Haddad has been promoted to the position of

Director of Mechanical Engineering. Haddad has been with the company for over 13 years and holds a BSME from Western Michigan University.



Valery Rudnev

Dr. Valery Rudnev has been promoted from chief scientist to Group Director, Science and Technology. Dr. Rudnev holds an MS in electrical engineering and a Ph.D. in induction heating. He holds 14 patents and has published 92 scientific engineering and research articles. He has been with the company for

over 6 years and in the induction heating industry for 24 years.



Peter Dickson

Peter Dickson, former manager of Research and Development, has been promoted to Director of Research and Development. He has been with Inductoheat for 15 years and holds a B.Eng. from Gippsland University in Australia.

Mihelick Nominated to Gear Research Institute Board

Joseph Mihelick, PE, has been nominated to the Gear Research Institute board of directors to fill the 2000-2002 Class A Trustee vacancy created with the departure from the board of Mr. Gary Kimmel. Mihelick is president of Gears Plus, Inc. and has 38 years experience in the development, design and manufacturing of enclosed industrial gearing. He is a mechanical engineering graduate of the University of Cincinnati and holds an MBA from Xavier University. Mihelick has three patents, has authored technical papers for both AGMA and ASME and has served on various gear and engineering-related boards. Gears Plus, Inc. is a consultancy that offers engineering services with an emphasis on gearing.

New Patent for Alfe Heat Treating

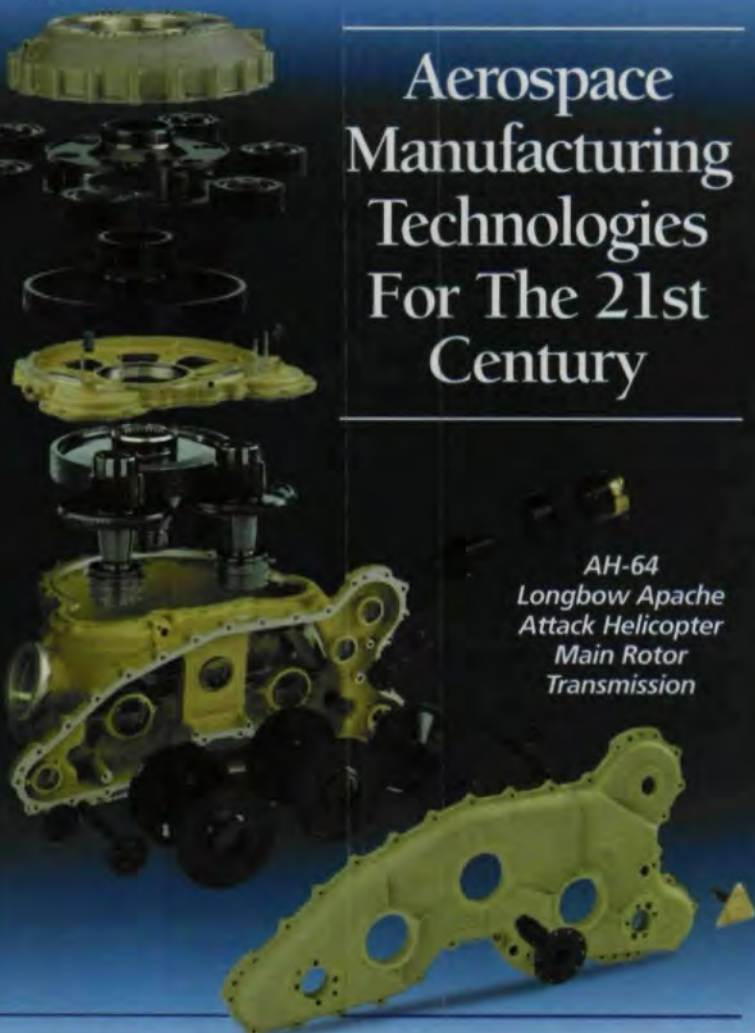
Alfe Systems, Inc., a division of Alfe Corporate Group, has received a patent for their new Double Level Furnace. The new heat treating furnace, described as a flexible, cost effective high production heat treating system, has multiple transport systems within the same furnace. This enables multiple heat treating cycle times using the same control temperatures to be run simultaneously. The furnace minimizes temperature variations by operating two discharge doors independently from each other. ⚙



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The Effect of Material Defects on Gear Performance—A Case Study

Raymond J. Drago, P.E. and Alejandro Font Filax, Ph.D.

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Abstract

The quality of the material used for highly loaded critical gears is of primary importance in the achievement of their full potential. Unfortunately, the role which material defects play is not clearly understood by many gear designers. The mechanism by which failures occur due to material defects is often circuitous

and not readily apparent. In general, however, failures associated with material defects show characteristics that point to the source of the underlying problem, the mechanism by which the failure initiated, and the manner in which it progressed to failure of the component.

In this case study, the authors examine the failure of a medium-sized pinion used in a mining application. The mode of failure was rather catastrophic in nature but did not follow any of the typically understood mechanisms such as tooth bending, surface distress, wear, etc. Often, as was the case for the subject pinion, material defects do not manifest themselves in these more typical tooth oriented failure modes, though the initial presentation of the failure often suggests more classical origins.

A complete shaft fracture was the ultimate cause for the pinion's removal from service. Initial inspection of the failed pinion indicated the presence of cracking in the toothed area. This cracking appeared to have progressed through the pinion and resulted in the shaft fracture, which caused the pinion to cease transmitting torque. In order to avoid a recurrence of the problem in this very critical application, it was very important to fully understand the failure including its cause and progression mechanism.

This paper presents a summary of the failure, its investigation, and the methods proposed for its resolutions. The data is presented in tutorial format so that the basic effect of the identified material defects can be better understood and used in future designs.

Introduction

In 1992, two new single helical gear sets were installed as part of a new mill system at a mine site. The pinion and gear sets are used to drive essentially identical ball mills used for processing ore from the mine.

After some relatively minor start-up problems, the gear sets were placed into normal service and

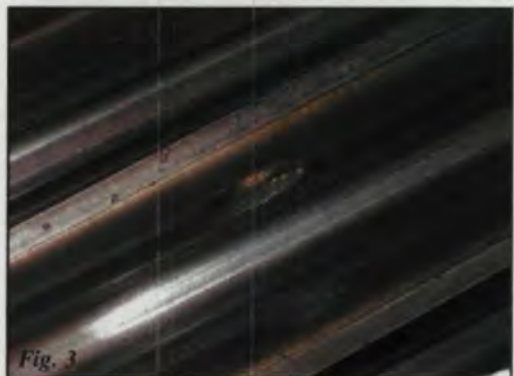


Fig. 1—Distress observed on ball mill pinion during 1994 visual evaluation.

Fig. 2—Close view of distress observed on ball mill pinion during 1994 inspection.

Fig. 3—Currently installed ball mill #2 pinion showing region (near mid-face) which has been ground out to remove pitting and surface cracks and then restarted.

performed without significant incident until some pitting was discovered during a visual examination conducted late in 1994. At that time, we noted only very minor surface distress, as Figures 1 and 2 show. The distress was located at about the middle of the face width and had been slightly ground out at the time of my November 1994 visit.

The pinion shown in Figures 1 and 2 was in service until its removal in August 1995 because of severe surface damage. It was reinstalled in mid-1998 and failed catastrophically in 1998, approximately three working years after start-up. The number of cycles accumulated by the pinion at the time of failure was lower than 3×10^8 . This, therefore, was certainly not a low cycle failure, but it was only one-third of the total design life.

The failed pinion was replaced and the mill restarted. During a visual inspection conducted in early 1999, the new replacement pinion was found to be exhibiting pitting very similar in both extent and location to the pitting in the original pinion. This pitting was hand dressed out and was found to have not progressed significantly during a visual inspection, which was conducted during April 1999, as Figure 3 shows.

During an inspection in April 1999, we did not observe any cracks on this pinion (we were unable to dye check it). However, the pinion was reported to have some cracks in the pitted region which were discovered during a previous dye check inspection. The cracks were ground out and the pinion was returned to service. At the time of our inspection, it appeared that only minor progression of the pitting was occurring.

The Second Helical Gear Set

As noted above, there are two identical mill gear systems on line at this site. The second mill was running at the time of our visit and we were unable to shut it down for a good look at the teeth. We examined the teeth through the inspection port with the aid of a strobe, which allowed the pinion and gear rotation to be "stopped" artificially. As Figure 4 shows, this pinion exhibited moderate surface distress in the form of pitting.

We could not detect any cracks on the surface of this pinion; however, it would be almost impossible to do so under the conditions of this inspection. The regions of pitting observed were similar in size to those observed on the first (failed) mill pinion, but they were in a somewhat different location along the face width. In the profile direction, the pitting was located near the pitch line on both mill pinions.

Lubricant distribution on both gear sets appeared to be adequate and uniform across the



Fig. 4—Pitting distress on second mill pinion teeth.



Fig. 5—Ball mill pinion #1 teeth, which exhibit no surface distress.



Fig. 6—Failed pinion removed from #2 ball mill in August 1998.



Fig. 7—Pinion fracture face. The light area at the top of the pinion is the crack propagation area. The dark area at the bottom of the pinion is the area that was cut after the failure.

face width. There was no indication of any lubrication-related distress on any of the pinions or gears examined.

The distress is located about 2/3 of the face width from the motor end of the face width. While Figure 4 shows typical distress, it is important to note that not all pinion teeth exhibit this distress. Figure 5 shows another group of pinion teeth on

Raymond J. Drago, P.E.

leads a double life. He is a senior technical fellow of the Boeing Company specializing in gear technology. He is also chief engineer of Drive Systems Technology, Inc., a gear engineering consulting company he founded in 1976.

Alejandro Font Filax, Ph.D.

is a professor with the Department of Mechanical Engineering at the University of Chile in Santiago.

Fig. 8—Pitting observed on fractured pinion. Note the ground out condition of the pitted area and the lack of new progression.



Fig. 9—Fractured tooth surface, suggestive of a subsurface flaw.



Fig. 10—Cracks progressed from the tooth roots.



the same pinion which exhibit virtually no distress at all.

The Failed Mill Pinion

The pinion, which failed in August 1998, was available for our visual examination; however, as Figure 6 shows, it was quite rusted. Still, our visual evaluation provided valuable clues regarding the cause of the failure.

It was apparent that a crack propagated through the pinion. The crack was on the divided face width at about the 2/3–1/3 point. The crack propagated through about 75% of the shaft thickness (Figure 7) but did not completely sever the pinion.

This failure was discovered during normal maintenance and no problem had previously been noted during operation. At shut down, maintenance personnel found a piece missing from one tooth and noted the crack extending about 75% around the circumference of the pinion.

It was reported that this pinion had been run in both directions and that the contact apparent on the tooth flanks confirms this. It was also reported to me that this pinion had been running in one direction only for "...many years..." The pitting observed was reported to have occurred generally at four locations, positioned about 90 degrees apart. The pitting which I observed on this pinion (Figure 8) was not sufficient, of and by itself, to account for this massive crack propagation failure. Note also that this level of pitting is not very different from that which I observed on this same pinion when I last visually evaluated it during my November 1994 visit (see Figures 6 and 7 above). Close examination of Figure 8 clearly shows that there was no real progression of the pitting distress and that the fracture does not appear to originate in the pitted region.

The surface characteristics of the failure, as shown in Figure 9, suggest the presence of a subsurface defect, which initiated the crack that propagated to failure. Despite the rusted nature of the fracture face, it is still relatively smooth in appearance. It is clear that multiple crack branches have propagated in the region of the fracture. This topography is suggestive of a fatigue crack propagation mechanism. Scanning electron microscopy of this region showed crack growth rates of 10^{-4} mm/cycle, that is, 80 mm in 3 days of operation.

Based on the view of the fracture face shown in Figure 10, it appears that the original failure had multiple origins and that it spread from the tooth root area down into the shaft section.

Current Pinion Condition

The conditions of the pinions and gears currently installed in both mills are very similar to each other. At this time neither pinion exhibits any critical problems; however, the overall condition of the pinions are very similar to the condition of the fractured pinion which failed in 1998. This is, of and by itself, some cause for concern.

The pitting which is apparent on these pinions is somewhat unusual in both its location and its appearance. The fact that the pitting is largely mid-face indicates that it will not be practical to realign the gear sets to favor the undamaged tooth surfaces. The appearance of the pitting suggests an over crowned condition on either the pinions or their mating gears or, perhaps, both. I have no indication of the actual geometry of these pinion and gear sets (i.e., lead and profile inspection charts or other similar measurements) which might identify the specific cause. Considering the experience of the failed pinion, however, and the similarity of the appearance of the pitting on all three pinions, it is

quite possible that the two currently installed pinions also exhibit some of the material anomalies which affected the failed pinion and, as described below, ultimately led to its fracture.

If this pitting is the result of material anomalies as was the fractured mill pinion described below, dressing the pitted area will lower the stress level in the region where the inclusions appear to be prevalent, thus reducing the possibility that they will cause a crack initiation and progression mechanism similar to that which led to the failure.

Failed Pinion Shaft Analysis

The fracture failure of this pinion shaft is an extreme example of the highly deleterious effects of material anomalies on the load capacity, reliability and life expectancy of large gears. The specific mode of failure experienced was via initiation of cracks near the top of one tooth with propagation occurring across approximately 75% to 85% of the shaft cross section, as Figures 11 and 12 clearly indicate. The cross hatched areas in these figures show the region which was cut to expose the fracture faces while the unhatched regions show the region over which the cracks progressed. Note especially the origin of the failure in the tooth top at the top of both figures and the extensive, branched crack network below the origin.

Despite this extensive propagation, no indication of the failure of the shaft was observed during normal operation of the mill. The fracture was only noted during a maintenance check. This crack propagation took place in less than three months of operation.

The condition of the tooth flanks and location of the fracture (almost mid-face) also indicated that the problem was not related to alignment. Further, the nature of the crack initiation and early progression indicates that the failure was not due to an overload condition, either one time or repeated. These factors point to problems with the material or manufacture of the pinion.

Based on the appearance of the fracture surface, a material anomaly is the primary suspect. In order to investigate this possibility, a detailed metallurgical evaluation was required.

The relatively good condition of the teeth over most of the face width combined with the large piece which was fractured away from one tooth near the middle of the face from just below the pitch diameter of the tooth tip very strongly suggests a localized anomaly. Pitting failures due to overloading or overall material insufficiency (e.g., low hardness, etc.) will manifest themselves relatively uniformly on all or most teeth while pitting failures due to misalignment-induced overloading

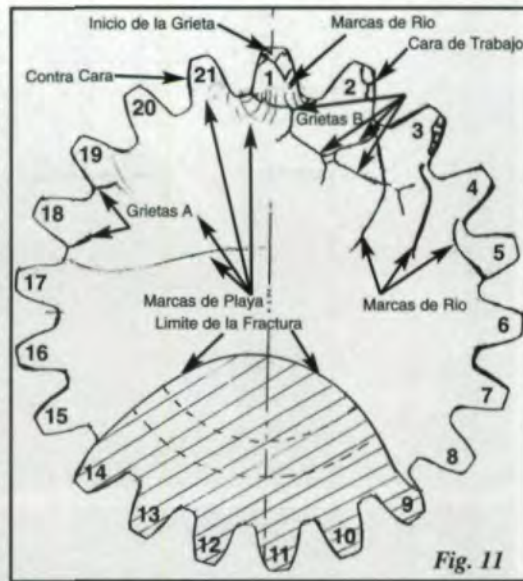


Fig. 11—Pinion fracture face. Mating surface shown in Figure 12.

Fig. 12—Pinion fracture face. Mating surface shown in Figure 11.

Fig. 13—Fractured pinion after removal. Note the large piece fractured from one tooth.

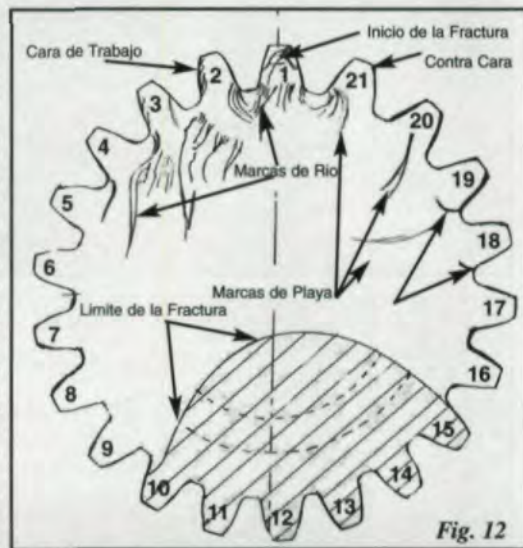


Fig. 12



Fig. 13

will occur in a localized region of the face width, biased to the more highly loaded end but also uniformly dispersed on most or all teeth. The surface failures observed on this pinion were highly localized, as Figure 13 shows.

The pinion was cut through the remaining shaft section to render the fracture faces visible. Figures 14 and 15 show the fracture faces after shaft sectioning. In Figure 14, the darker region in the upper right of the photograph is the section that was cut through to reveal the fracture face, which

Fig. 14—Pinion fracture face corresponding to Figure 12.



Fig. 15—Pinion fracture face corresponding to Figure 11.

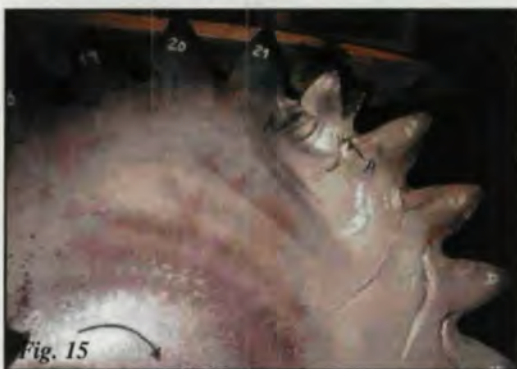


Fig. 16—Inclusions that were found in the fracture origin region.

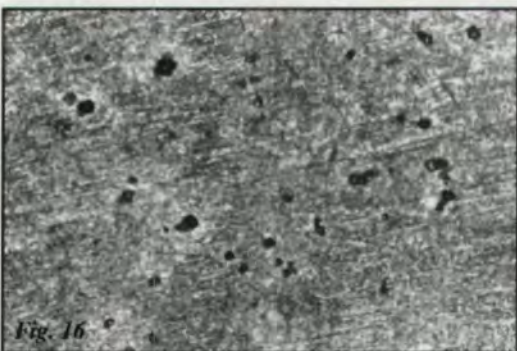
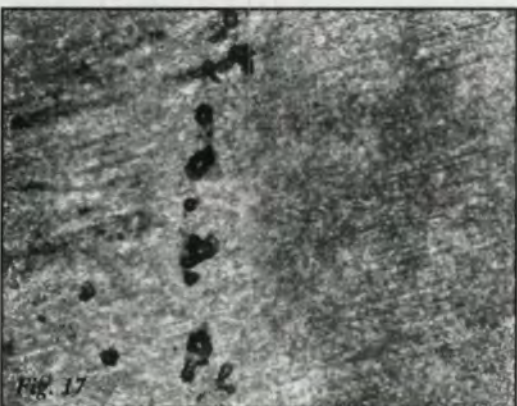


Fig. 17—Inclusions clustered in a longitudinal pattern.



is the smoother surface at the lower left of this photograph. It is clear from these figures that the origin of the crack network is in the tooth region identified by the arrows shown in Figure 15.

Metallurgical Evaluation

In order to better understand the nature of the fracture, metallurgical samples were obtained

from the origin region and subjected to detailed analysis. Based on this evaluation, the specific mechanism of crack initiation became quite clear.

Once polished and etched, these sections revealed the presence of large concentrations of inclusions, as Figure 16 shows. These inclusions were widely distributed in the region where the cracks originated. Such inclusions are, of course, impurities in the basic steel forging. The effect of inclusions on the capacity of the gear depends very much on their location.

Inclusions act as stress concentration factors which effectively increase the nominal, load induced stress level. Depending on their location on the tooth and the stress field which exists at that point, such inclusions may prove harmless or they may cause substantial localized stress concentrations, which can result in crack initiation. The length of time required for crack initiation and progression depends on many factors including the magnitude of the basic applied stresses. If the basic applied stresses are very high and concentrated in a single region, such as tooth root bending stresses, the crack may initiate and propagate quickly. Conversely, if the basic applied stresses are more diffuse, such as they are in the contact region, cracks may not initiate for a very long time and, once initiated, may also have long propagation times. The latter appears to be the case here.

Inclusions were found throughout the origin region, but the ones which most directly influenced this failure are those which were found close to the surface of the tooth. If the inclusions are well below the tooth surface and widely dispersed, their effect may be small. When close to the surface or in the tooth fillet region, however, they can be quite significant. Concentrations of inclusions can also pose a very great problem for carburized pinions when they are heat treated. This pinion was through hardened, thus this aspect was not observed.

Of even greater significance is the situation which occurs when inclusions occur in clusters, as Figure 17 shows. These clustered inclusions form a very large stress concentration and can adversely affect both the surface load capacity and bending fatigue strength of the teeth. When clustered, the negative effect of an inclusion is magnified many times over.

It was a group of inclusions, such as the ones shown in Figure 17, which caused the failure of this pinion. Specifically, the clustered inclusions were located below the surface but within the high stress region. As Figure 18 shows, the fracture passed through the pitted region. The pitted

region itself, however, did not propagate in any significant manner as is also obvious from Figure 18. Early in the life of the part, inclusions contributed to the occurrence of the odd pitting pattern, which was originally ascribed solely to over crowning. It is likely that over crowning may also have played a role by further concentrating the stresses in this weakened area; however, it is not possible to determine with certainty what the proportional effects were. It is likely, however, that the inclusions were the major factor.

In Figure 18, note that the tooth before and the tooth after the pitted tooth in the center of the photograph are both essentially free of pitting. Since crowning (normal or excessive) would be essentially uniform on all teeth but the distribution of material defects such as inclusions is always nonuniform, the lack of distress on some teeth indicates that the inclusions are, in fact, the driving factor.

This pitting, combined with the presence of other inclusions, in the presence of the applied, repetitive surface stresses caused cracks to initiate below the tooth surface. This same loading pattern caused the cracks to progress, very slowly at first (this early manifestation was the pitting pattern observed early in the life of the gear). As the crack length grew, the critical stress threshold also dropped, thus the cracks began, eventually, to propagate more rapidly. Eventually, they propagated through the shaft section, resulting in the catastrophic failure observed.

Failure Scenario

Regarding this pinion's specific failure scenario, it is clear that the cracks which propagated through the shaft originated in the teeth just below the surface due to inclusions, which acted as nucleation points due to their stress concentration effects. The inclusions are an indication that the steel used in these pinions was not of adequate cleanliness for this application. In addition, as described above, because of the appearance of pitting, which is similar to that observed on the failed pinion, the pinions currently installed in the mills are also of some concern in that the same conditions which produced the failure of the first pinion may possibly also exist in these pinions as well.

Ultrasonic Inspection

The presence of inclusions in this pinion also indicates that the raw material (forging) used was either not ultrasonic inspected (UT) or that the inspection was not properly evaluated. Such inclusions can be successfully detect-

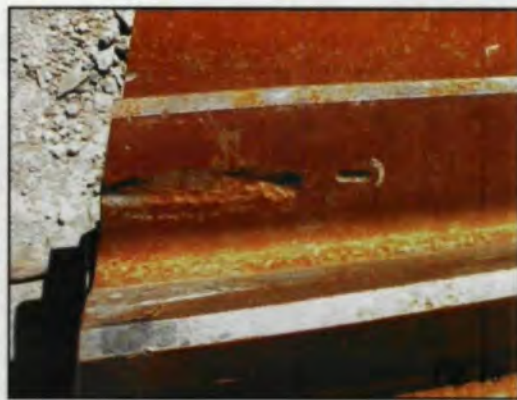


Fig. 18—Crack passed through pitted region. Note that the teeth before and after the pitted tooth are both free from pitting.

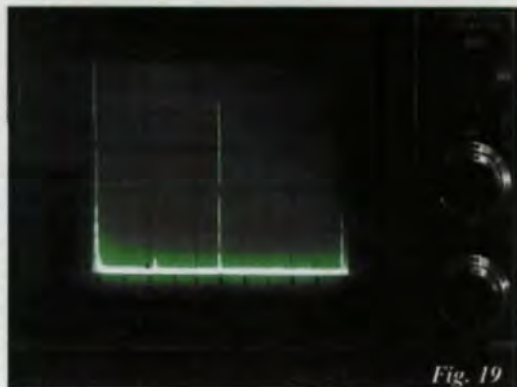


Fig. 19

Fig. 19—Ultrasonic indication of subsurface defect in a gear forging.

ed by a properly conducted UT inspection. For example, Figure 19 shows the result of a UT inspection of another forging which did exhibit a subsurface indication. It is important to note that this figure is provided for demonstration purposes only as this forging was not evaluated metallographically to define the cause of the indication. UT inspections should always be required for all critical pinion and gear blanks.

Conclusion

Material defects can have a very significant detrimental effect on the long term viability of critical gears. These detrimental effects may occur in a relatively short period if the applied stress levels are high but in the presence of more moderate stresses, failures may occur after a very long period of very successful operation.

Recommendation

All critical forgings should be ultrasonically inspected to detect subsurface material defects before detail machining is started. In addition, material specifications for such gears should include cleanliness requirements and a definition of testing which will confirm the cleanliness rating. The use of vacuum degassed materials will also help to minimize the chance of a subsurface defect adversely affecting the load capacity or life of the finished gear. ⚙

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Ferritic Nitrocarburizing Gears to Increase Wear Resistance and Reduce Distortion

Loren J. Epler

Quality gear manufacturing depends on controlled tolerances and geometry. As a result, ferritic nitrocarburizing has become the heat treat process of choice for many gear manufacturers. The primary reasons for this are:

1. The process is performed at low temperatures, i.e. less than critical.
2. The quench methods increase fatigue strength by up to 125% without distorting. Ferritic nitrocarburizing is used in place of carburizing and hardening, carbonitriding, nitriding or in conjunction with conventional and induction hardening.
3. It establishes gradient base hardnesses, i.e. eliminates egg-shell effect on TiN, TiAlN, CrC, etc.

In addition, the process can also be applied to hobs, broaches, drills and other cutting tools.

History. Ferritic nitrocarburizing was first established in 1947 in a cyanide, salt-based batch process. It was later refined



Fig. 1—.0008" Compound of White Layer on an SAE 1035 Gear, Rockwell C Hardness 75+.

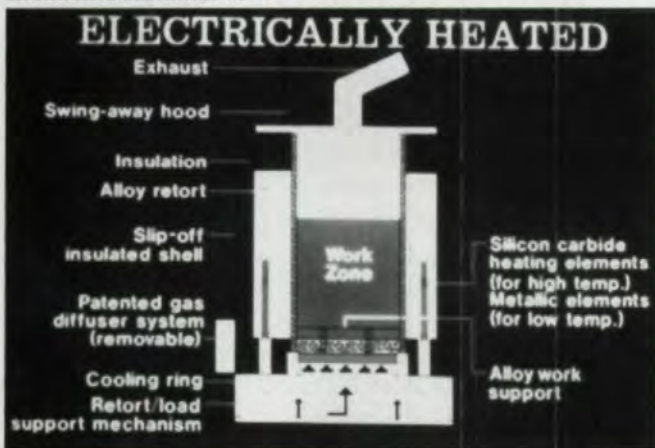


Fig. 2—Fluidized bed furnace.

as a gaseous ferritic nitrocarburizing process and patented by Lucas Industries in 1961. Lucas demonstrated that they could produce surface layers identical to those produced in salt bath processes using an endothermic, ammonia-based atmosphere.

The process was classified as a "thermochemical surface treatment" that involved the diffusion of both nitrogen and carbon into the surface of a metal at a temperature below the austenite transformation temperature. The process would yield a single phase epsilon layer with an atomic weight of Fe_2O_3 . The single phase layer makes the product much more wear resistant than gas or ion nitriding, according to Dawes and Trantner (1).

In 1982, Ironbound Heat Treat developed Nitrowear® using similar atmospheres in a fluidized bed medium. Subsequently, Jack Ross, owner and founder of Ironbound, patented and licensed the process to Dynamic Metal Treating. The fluidized bed proved to be up to six times faster than the conventional atmosphere furnaces with less temperature variation, typically $\pm 2^\circ F$ from the process setpoint. Ferritic nitrocarburizing and its corrosion resistance process, Nitrowear, are similar to ion and gas nitriding but produce less brittleness and greater wear resistance.

Since taking on the license, Dynamic Metal Treating has developed and refined the process and has many of its own patents. Dynamic currently has over 25 derivatives of the process, engineered to meet the product's application and environmental requirements. The process is applied to AISI/SAE 4140, 4150, 5160, 52100, 8620, 12L14, 1144, D-2, H-13, M2, M4, M42 and T-15, just to name a few.

The Ferritic Nitrocarburizing Process

The process temperatures for ferritic nitrocarburizing are anywhere from $700^\circ-1,200^\circ F$, depending upon the base material and the desired metallurgical properties. Since the temperature is less than the base material's critical temperature, typically $1340^\circ F$ for structural and alloy steels, the parts do not experience any distortion when quenched.

During processing, nitrogen, anhydrous ammonia and a carbon producing atmosphere are introduced to the material. Based upon time, temperature and the ratio of these gases, two metallurgical characteristics are diffused (become part of the base metal) into the metal. They are referred to as:

1. Compound layer (also known as white layer or iron epsilon layer), and
2. Diffusion zone (also known as carbon and nitrogen enriched zone or total case).

The white layer has a Rockwell C hardness of 75 and greater, and the diffusion zone can be anywhere from HRC 21 to 70, depending upon the base metal. As a result, the product has:

- High wear characteristics and lower coefficient of friction values,
- Corrosion resistance greater than or equal to stainless steels,
- Low distortion, i.e. no lead, involute, tooth spacing or pitch diameter runout changes and .0002" per side linear growth on measurements over balls.
- Anti-spalling and galling properties.

These characteristics also make this a viable process for cutting tools such as hobs, broaches, etc. and an alternative to titanium nitride or other comparable coatings at one-quarter the cost. Ferritic nitrocarburizing can be applied to all ferrous metals and 400 series stainless steels. When used with precipitation hardening, special surface activation is required.

Fluidization. Fluidization is the technique where aluminum oxide (sand) behaves like a liquid while in a heat treat furnace. This is achieved by introducing gases at a given flow rate and pressure through a diffusion plate and into the sand. The diffusion plate is primarily a distribution plate that causes the pressurized sand and gases to react and become microscopically separated, thus causing the medium to move freely throughout the furnace shell. The furnace is heated electrically with silicon carbide heating elements, which are located outside the heat treat vessel's shell. The heat is rapidly distributed throughout the work zone, thereby attaining excellent furnace temperature uniformity and rapid heat transfer into the workpiece being processed. The temperature is being controlled to within $\pm 2-5^{\circ}\text{F}$ while conventional pit nitriding experiences $\pm 10-25^{\circ}\text{F}$. On pre-heat treated materials, the ferritic nitrocarburizing cycle temperatures can mirror conventional tempering values.

The surface of the part being treated is constantly scrubbed with fresh, reactive gases. The fluidized bed methods are not inhibited by part geometry or blind holes, thus ensuring uniform metallurgical properties regardless of the part's configuration. The aluminum oxide sand particles are nonabrasive and often enhance microfinishes (see Table 1).

Before Ferritic Nitrocarburizing Ra	After Ferritic Nitrocarburizing Ra
< 4 Ra	4-8 Ra
> 4 Ra but < 16 Ra	4-12 Ra
> 16 Ra but < 32 Ra	14-28 Ra
> 32 Ra but < 125 Ra	30-125 Ra

Table 1—Changes in microfinish with the ferritic nitrocarburizing process.

Gear Type	Application
Helical, ductile iron balance gear	Balance gear, 4-cylinder engine
Splined, 8620 gear	Transfer case, 4-wheel drive
4140 and 8620 spur gears	Rear wheel drive applications
H-13 drive gears	Steel mill cranes
1035 gear	Clutch hub, transmission

Table 2—Gear types and applications that could benefit from the ferritic nitrocarburizing process.



Fig. 3—Gears and other components treated with ferritic nitrocarburizing.

The fluidized bed furnace can also perform many other heat treat processes such as nitriding, hardening (up to 1,950°F), steam bluing, annealing, stress relieving, carburizing and carbonitriding.

Gear Applications. The useful life and warranty period for many gears can be extended with ferritic nitrocarburizing. Some of the gear types and applications are listed in Table 2.

Additional benefits for gears include temper resistance (resistance to softening from heat generated in service), increased lubricity at cold start and increases in the tensile, yield and fatigue strength of the base metal as well as high compressive stresses (229,000 KSI). The process can also increase throughput and eliminate problems such as bell-mouthed I.D.s, quench cracking, retained austenite and bainite, and decarburization.

Steel Selection. Gear steels that have been successfully processed using ferritic nitrocarburizing are: 1008-1020, 12L14, 1215, 1141, 1144, 8620, 4140, 4150, 4340, ductile iron and pearlitic-malleable. The low carbon and leaded steels are selected when the application is under low, applied torsional and axial loading needing primarily wear resistance. The 11XX series is used for its increased machinability while maintaining some core hardness properties. Higher torsional and axial loading or where backlash occurs may require higher alloyed materials so that the core properties can be maintained during ferritic nitrocarburizing.

Case Depth/Quality Requirements. Quality assurance of ferritic nitrocarburizing is typically very simple. Parts are checked for file hardness per SAE J864, while white layer and diffusion zones are verified through metallography.

File hardness values for low carbon steels are typically reflected as 58 minimum with the value increasing according to the base material. White layer depths at 1000 X are anywhere from .000050-.0022" (.00127-.056 mm) and diffusion zones from .0013" to > .032" (.033-.81 mm). ◉

Loren J. Epler is president of Dynamic Metal Treating, Inc., of Canton Township, MI. This article is based on the presentation he gave at the SME Heat Treating and Hardening of Gears Clinic that was held on October 25, 1999 in Nashville, TN.

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Reducing Production Costs in Cylindrical Gear Hobbing and Shaping

Prof. Dr.-Ing. Fritz Klocke and Dipl.-Ing. Claus Kobialka

Introduction

Increased productivity in roughing operations for gear cutting depends mainly on lower production costs in the hobbing process. In addition, certain gears can be manufactured by shaping, which also needs to be taken into account in the search for a more cost-effective form of production.

One way of increasing the productivity of the hobbing process is to raise cutting speed. Another potential strategy for reducing costs and thus increasing productivity is to eliminate the use of cooling lubricants. High-speed hobbing is now understood almost exclusively as dry machining with carbide tools at cutting speeds in excess of 300 m/min (Ref. 2). Together with the introduction of such new technology, productivity can also be enhanced through the use of innovative high speed steel cutting tools.

The potential of each of these rationalization measures (use of HSS and carbide tools) is discussed below for the case of hobbing. This information is supplemented by an indication of the technological limits and process reliability of specific applications. Finally, because certain gear cutting applications require the use of shaping, a lubricant-free option for manufacturing internal gears is indicated.

In order to make a comprehensive and reliable assessment of the required productivity, the report first analyzes wear-relevant mechanisms in hobbing and shaping operations. Various tool materials and coatings are investigated at differing cutting parameters in relation to the special stresses encountered in machining at high specific removal rates and differing machining conditions. The objective is to demonstrate both the performance potential and the current limits for existing tool systems. Finally, the tool lives determined in the tests are assessed.

State of the Art

Economic and technological significance of cooling lubricants. Modern cooling lubricant systems make a decisive contribution to the high level of performance of numerous production processes by performing their main tasks of cooling and lubricating the contact point and by transporting chips away from it. Despite their great technological importance, they have been the target of increasing criticism in recent years, stimulated by rising operational and disposal costs due to stricter environmental regulations.

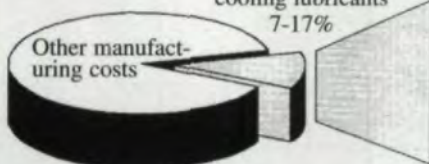
Consideration of these problems has made many users aware of the costs involved in the use of cooling lubricants. A recognition that part-related costs for the cooling lubricant system may be several times higher than tool costs has led to a reevaluation of cooling lubricant use in many companies. The logical conse-

Ecological reasons:

- Pollution due to disposal
- Oil mist
- Stricter laws & regulations

Economical reasons:

Costs for using cooling lubricants
7-17%



- Cooling lubricant facilities
- Exhaust-type mist collector
- Purchase of the lubricant
- Maintenance and process costs
- Disposal costs
- Loss with parts
- Other costs

Medical reasons:

- Skin exposure to chemicals
- Inhalation of oil vapor

Fig. 1—Reasons for eliminating cooling lubricants (Ref. 7).

quence is a demand for solutions which reduce or eliminate the use of cooling lubricants and hence the associated costs (Fig. 1, Ref. 1).

Especially in terms of economic criteria, purchase costs for the cooling lubricant must be seen in association with investment costs for a cooling lubricant unit. Cooling lubricants also need maintenance and disposal (Refs. 2-6). Losses on parts and chips, vaporization and evaporation likewise contribute to lubricant consumption. Overall, costs to the company associated with the use of cooling lubricants represent an increasing economic burden and amount to some 7% to 17% of proportional part costs in the automotive industry. Calculations for gear making indicated that between 16% and 30% of hobbing costs result from the use of cooling lubricants. In view of restrictive environmental legislation reshaping regulations for the disposal and reutilization of special wastes, it may be anticipated that lubricant-related costs will rise even further.

Potential of new or optimized gear cutting tools. Together with cost reductions through the elimination of cooling lubricants, it is possible to increase productivity by using higher cutting parameters. Apart from innovative tool materials, tool coating with hard, thin films should not be neglected in this respect.

PVD coating technology in particular opens the way to improved cutting parameters as compared to the use of uncoated tools in machining technology. Coating technology has consequently undergone rapid development in recent years. The results of this development include new hard, thin-film systems that can be deposited with consistent results on tools of any degree of complexity. The effect has been to make the use of fully coated complex tools, especially hobs, state-of-the-art technology (Ref. 8).

Current commercial systems include TiN, Ti(C,N), (Ti,Al)N and other hard, thin films based on the elements titanium and aluminum, which can be deposited either on high-speed steel or on carbide. Because of their wide availability and differentiated mechanical properties, TiN and (Ti,Al)N coating systems will be considered in greater detail below.

Machining Tests

Tool wear behavior and tool lives for hobbing operations with HSS tools using cooling lubricants. All tests were carried out with a maximum crest chip thickness according to Hoffmeister (Ref. 9) of $h_{cu\ max} = 0.18$ mm. Cutting speed was increased in 30 m/min stages

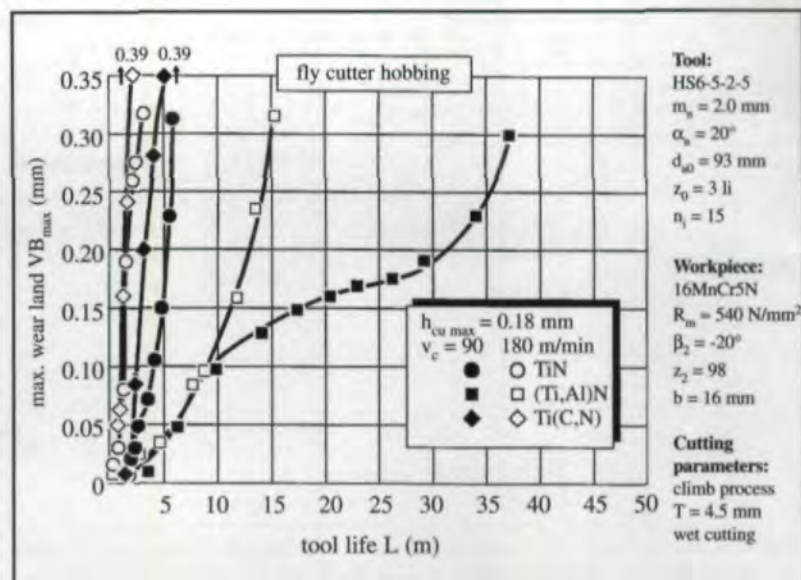


Fig. 2—Wear curve at different cutting speeds.

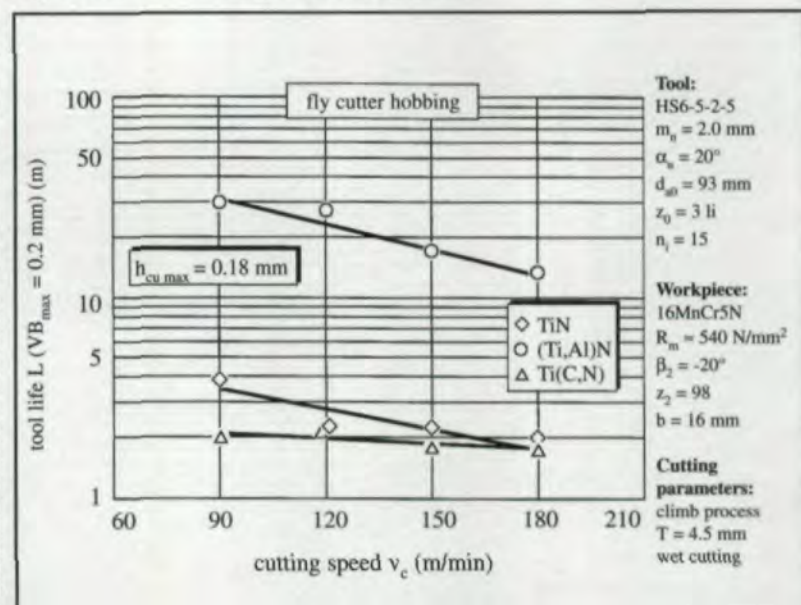


Fig. 3—Comparative tool life as a function of cutting speed.

from $v_c = 90$ m/min to $v_c = 180$ m/min. The width of wear land versus tool life for cutting speeds from $v_c = 90$ m/min and $v_c = 180$ m/min shown in Figure 2 are representative for all analyzed cutting parameters.

Initially, the tool life criterion aimed at was a maximum face wear of $VB_{max} = 0.3$ mm. The measured wear curves indicate, however, that a tool life criterion of $VB_{max} = 0.2$ mm is more effective in reducing progressive wear (Fig. 2). The tool life comparison is therefore presented for a tool life criterion of $VB_{max} = 0.2$ mm (Fig. 3).

Shorter tool life was realized as cutting speed was increased from $v_c = 90$ to $v_c = 180$ m/min, irrespective of the coating system. The reason for this is the increased thermal stress on the tool caused by the high relative velocities of the tribological partners and the corresponding

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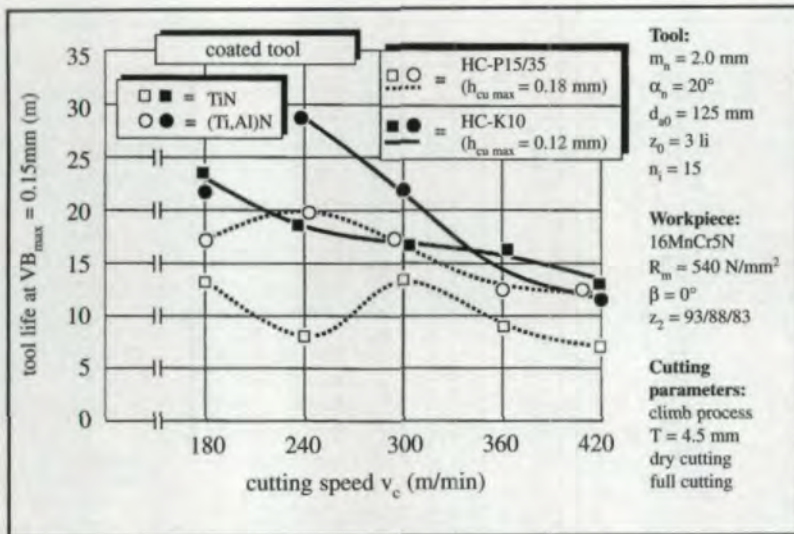


Fig. 4—Tool life for fully coated carbide tools with tool-life-optimized crest chip thicknesses and a maximum width of wear land $VB_{max} = 0.15$ mm.

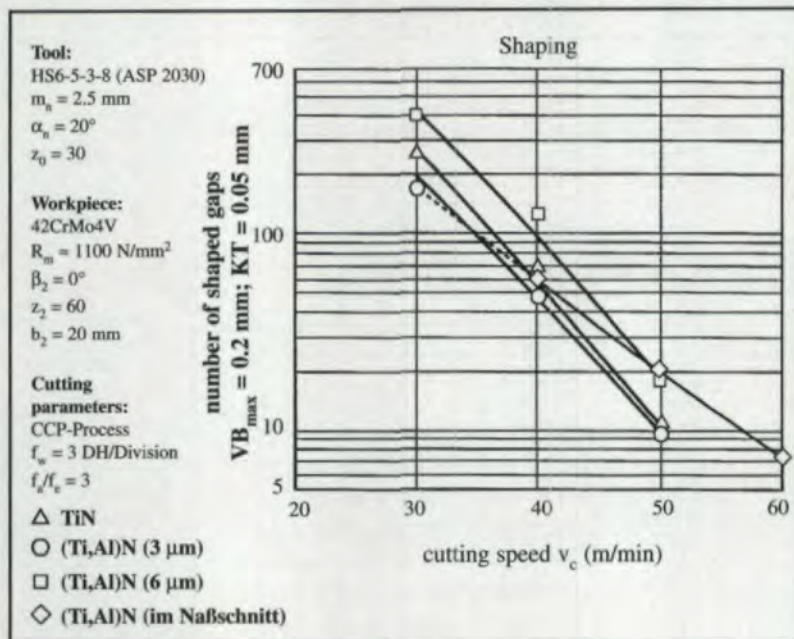


Fig. 5—Comparative tool life as a function of cutting speed.

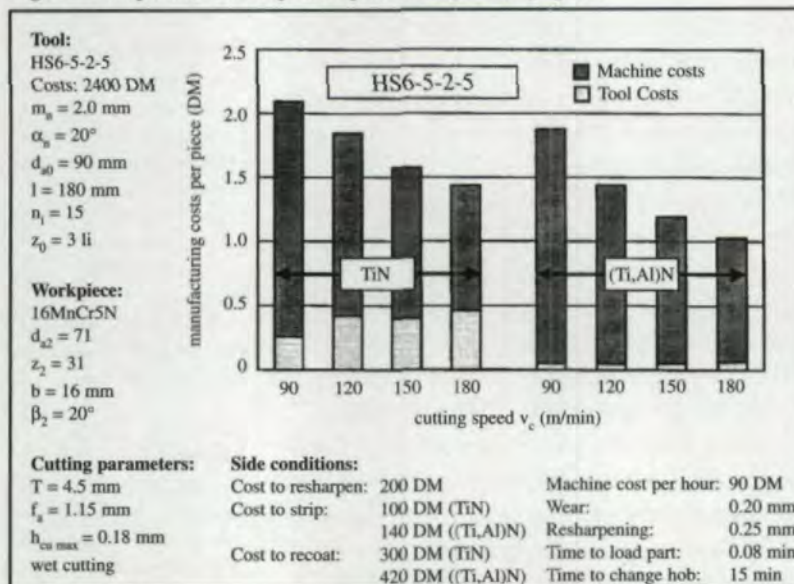


Fig. 6—Production costs as a function of coating.

increase in frictional power. The relatively long tool life seen at a cutting speed of $v_c = 180$ m/min is nevertheless surprising. The exceptional wear resistance of HSS tools at high cutting speeds may be attributed partly to the insulating effects of the hard, thin film and partly to the shortened chip length in hobbing a workpiece with a width of 16 mm. A calculation of the maximum chip length according to Hoffmeister (Ref. 9) indicates chip lengths of 19.5 to 21.4 mm as a function of the feed rate. This maximum chip length is not reached, due to the workpiece of 16 mm to be machined helically with $\beta = 20^\circ$. The thermal stress on the high-speed steel during hobbing of a narrow gear is lower than for solid hobbing of a broader gear.

Whereas tool lives achieved with (Ti,Al)N-coated tools were on average higher than those for TiN-coated tools by a factor of roughly 7, the tool lives of Ti(C,N)-coated tools were even shorter. Almost irrespective of cutting speeds, the maximum attainable tool life for Ti(C,N)-coated tools is approximately 2 m.

Dry hobbing with carbide tools. Carbide hobs have been available on the tool market as an alternative to HSS hobs for a number of years and have been used increasingly in series production since the introduction of hobbing machines designed for dry cutting. Studies of the wear behavior of coated carbide hobs reveal face wear versus tool life trends identical with those already shown for HSS tools in Figure 2.

On the basis of thermal emissions generated by the machining process and proportional to the maximum width of wear land, industry uses carbide tools up to a maximum wear land width of $VB_{max} = 0.15$ mm. At larger wear land widths, the generated heat may also affect the fixture during a dry cutting operation, leading to fixing problems for new blanks.

Studies on coated carbide tools have shown that maximum tool lives consistent with reliable process behavior are achieved with carbides in applications group P15/35 at a maximum crest chip thickness of $h_{cu,max} = 0.18$ mm. Axial feeds in which a maximum crest chip thickness of $h_{cu,max} = 0.12$ mm are realized are advantageous in the case of applications group K10 carbides.

Against this background, a comparison of test results for wear land width $VB_{max} = 0.15$ mm and optimum substrate-dependent crest chip thicknesses with HC-P15/35 and HC-K10 carbides is of interest (Fig. 4).

Longer tool lives are achieved with (Ti,Al)N-coated tools than with TiN-coated tools on iden-

tical substrates. In the range of cutting speeds of interest for dry cutting ($v_c \geq 300$ m/min), longer tool lives are achieved with HC-K10 carbide substrates than with HC-P15/35 equivalents. The number of feed markings on a tooth flank must, however, be taken into account separately for each specific application in terms of subsequent production processes for a gear. If, for example, a subsequent shaving process is envisaged in the process chain, values may fall neither above nor below a range of feed markings on the tooth flank, and this fact must be taken into account in selecting the tool substrate or geometry.

Tool lives in shaping. Shaping comes second only to hobbing as a significant production process for the generation of cylindrical parts. Coated HSS tools are mainly used for shaping, because mechanical machine concepts allow high cutting speeds to only a limited extent, and the use of cost-intensive carbide tools for soft cutting does not appear cost-effective. Shaping is used wherever the penetration resulting from the tool and workpiece geometries cannot be generated by a hob. A characteristic example is the generation of internal gear teeth.

Figure 5 summarizes the tool lives that can be achieved with variously coated tools as a function of cutting speed for the case of highly-tempered internal gears. The tool life curves can be approximated very closely by straight lines in the logarithmic presentation.

Tool lives vary inversely with rising cutting speed with all coating systems in the tests. This phenomenon may be attributed to the sharply increasing thermal stress on the tool as the cutting speed rises. Only very slight differences in tool life are observed for all coating systems in the high cutting-speed range. The wear behavior of the tools may be regarded as process-reliable in all the cutting parameter fields shown in the figure.

Of interest is the significant lengthening of tool life when the coating thickness is increased. Doubling the film thickness from $3 \mu\text{m}$ to $6 \mu\text{m}$ virtually doubles the number of gaps which can be shaped. Approximately 380 gaps were machined at $v_c = 30$ m/min, corresponding to a tool life length of roughly 7.5 m. In the logarithmic presentation, increasing the coating thickness leads to a parallel shift in the wear curves.

An increase in tool life of some 20 machined gaps per tool tooth as compared to dry cutting methods is achieved by using cooling lubricant. The tool life curve also climbs less steeply when cooling lubricant is used.

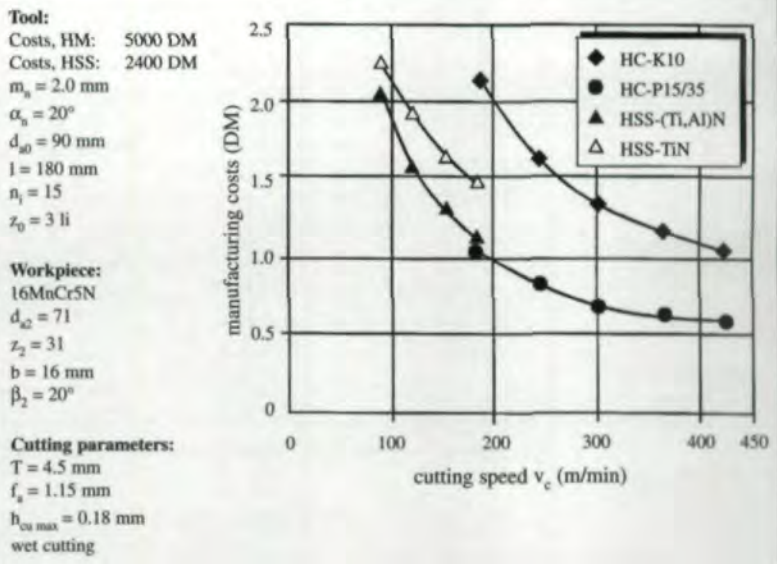


Fig. 7—Part costs in hobbing as a function of cutting speed.

The effects of the selected feed rate on the wear behavior of the tools will not be discussed in greater detail. It may, however, be stated that a reduction in tool life results from an increase in the feed rate, irrespective of the coating. The decrease in tool life is, however, smaller than from an increase in cutting speed. These data have already been described (Ref. 10) for the shaping of 16MnCr5 steel cylindrical gears.

Economic Analysis of Tool Lives

A knowledge of attainable tool life is not in itself sufficient to justify the use of a specific tool system. Of greater interest for the industrial user is a knowledge of part-related production costs. In this section, the technological test results presented above are analyzed for a fictional case to determine the production costs resulting from the tool life.

A machine hourly rate of 90 DM (approximately \$60) was assumed for hobbing or shaping operations, rising to 100 DM (approximately \$65) per hour if cooling lubricant is used. The tool costs presented in the following figures are based on current cost structures for medium-sized tool batches of roughly 10 to 20 tool inserts per month. (Ti,Al)N coatings are more expensive than TiN coatings, and a cost supplement of 40% was calculated for coating and stripping.

Hobbing. Analysis of the production costs per finished gear using HSS tools indicates lower costs at higher cutting speeds both for TiN- and (Ti,Al)N-coated tools (Fig. 6). The analysis is based on the constraints described in the figure and on the tool lives determined as a function of the coating system and cutting parameters for the case of fly cutter hobbing.

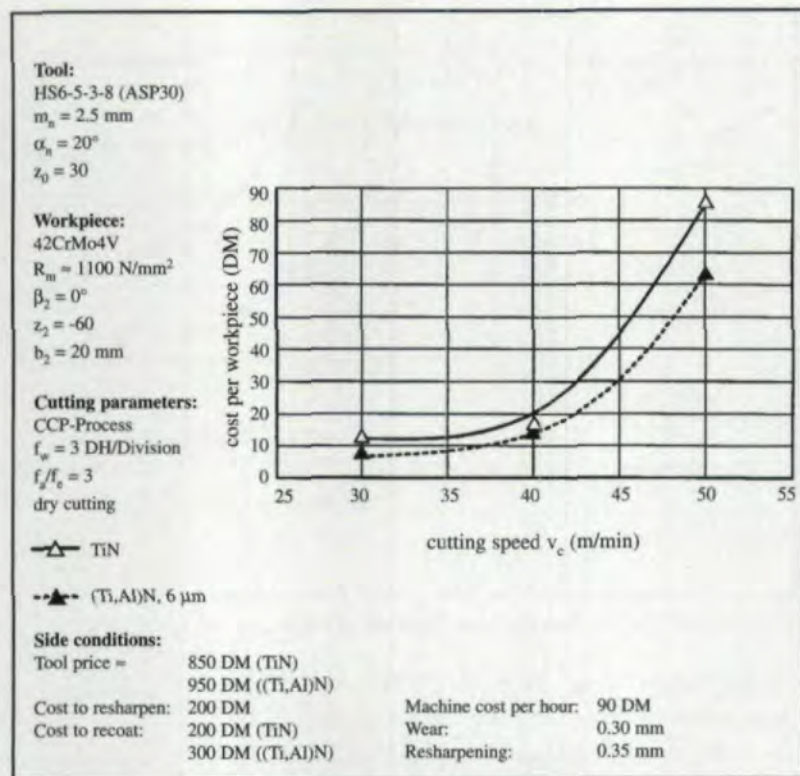


Fig. 8—Part costs for shaping as a function of cutting speed.

The pure production costs take into account the hourly machine rate, the tool purchase costs and the tool conditioning costs (regrinding, stripping and recoating).

Two cost trends emerge with rising cutting speed. On the one hand, the machine costs per part diminish, since more pieces are produced over the same period when the cutting speed is increased. On the other hand, the tool costs per part increase since tool lives shorten as the cutting speed rises, and fewer pieces are produced for each regrinding of the tool. Cumulatively, however, the decrease in machine costs outweighs the increase in tool costs.

Taking hobbing with TiN-coated HSS tools at a cutting speed of $v_c = 90$ m/min as a basis, an increase in cutting speed to 180 m/min results in a cost saving of roughly 15%. This benefit results from the 41% fall in machine costs, although tool costs rise by 100%.

The tool cost component of production costs per part for hobbing with (Ti,Al)N-coated tools at a cutting speed of $v_c = 90$ m/min amounts to no more than 3%; at $v_c = 180$ m/min, the corresponding figure is 11%. This means that potential cost reductions should be sought mainly by shortening machining times and diminishing machine costs, and less by reducing tool costs. If part quality permits an increase in the feed rate from $f_a = 3$ mm to $f_a = 5$ mm, the resulting increase in maximum crest chip thickness $h_{cu \max}$

from 0.18 mm to 0.24 mm may be used to exploit an additional cost reduction potential through low tool costs and machine costs.

Taking hobbing with TiN-coated tools at a cutting speed of $v_c = 120$ m/min as standard practice in this module range, there is a saving in production costs per part of roughly 47% with (Ti,Al)N-coated HSS hobs and a cutting speed of 180 m/min.

The influence of cutting speed on the production costs determined for HSS tools in the test is also applicable to carbide tools. The basis of calculation is a 10% lower machine hourly rate for dry as opposed to wet cutting. This is due to the elimination of cooling lubricant costs as discussed in Figure 1. Because TiN-coated carbide tools are less expensive than (Ti,Al)N-coated tools, but also attain shorter tool lives, an analysis of the results shown in Figure 4 established no significant difference in production costs between TiN- and (Ti,Al)N-coated tools at the same cutting speed (Fig. 7).

The analysis was based on a 40% higher cost component for coating with (Ti,Al)N as opposed to TiN. In the future, when (Ti,Al)N coatings are in more widespread use, costs are likely to be more favorable.

At the same cutting parameters, a reduction in production costs may be expected simply from the increase in cutting speed (see Fig. 7) because machine costs are such a dominant factor. The increase in tool costs with rising cutting speed is marginal compared with the potential machine cost savings.

The same result is evident if production costs for HSS tools and K10 carbides are compared. Owing to the small chip thicknesses, which can be achieved with HC-K10, a higher gear-cutting rate is possible at a lower feed rate and identical production costs of roughly 1.10 DM per part, because the cutting speed is more than doubled from 180 m/min to 420 m/min. At the same time, a lower feed rate improves part quality due to the lower kinematic roughness.

Shaping. Unlike hobbing, production costs for the shaping of highly-tempered work materials are dominated by disproportionate tool wear and the resulting tool cost component. Coating costs are again assumed to be 40% higher for (Ti,Al)N as opposed to TiN. Because of the technological advantages of a (Ti,Al)N coating, however, a reduction of between 25% and 50% in production costs is feasible. The higher coating costs are outweighed by the longer tool lives (Fig. 8).

Conclusion

It has been shown that improved performance and cost savings of roughly 50% can be achieved with coated HSS tools in a wet cut on conventional hobbing machines if (Ti,Al)N coatings are used and removal rates are adapted accordingly.

The same trends have been demonstrated for the dry cutting of highly-tempered internal gears using a shaping process with HSS tools. Although a higher cutting speed can be obtained with cooling lubricants, the tool life achieved makes dry cutting an ecologically and economically interesting alternative to conventional machining strategies with cooling lubricants. The basic precondition for eliminating cooling lubricants is, however, the use of coated tools.

Cylindrical gear production with carbide tools in a hobbing process reduces production costs simply through shortened machining times, although tool costs are significantly increased. The technological advantage of a (Ti,Al)N hard coating is not mirrored in lower part-related production costs, owing to higher coating costs as compared to the TiN coating. Savings are rather to be found in higher cutting speeds, which can be realized through the hard coating system, allowing reduced machine and production costs.

On both HSS and carbide tools, these cost savings are generated partly by significantly increased tool life, as opposed to conventional TiN coatings, with a corresponding reduction in the tool cost component; and partly through the faster cutting speeds allowed by the lower abrasive wear and greater high-temperature hardness of the coated tools. ◉

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Microsecond Heat Treatment of Gears

Dale C. McIntyre, Kerry P. Lamppa, Macie M. Sturm,
Eugene L. Neau, and Regan W. Stinnett

The performance of metal surfaces can be dramatically enhanced by the thermal process of rapid surface melting and re-solidification (RMRS). When the surface of a metal part (for instance, a gear) is melted and re-solidified in less than one thousandth of a second, the resulting changes in the material can lead to:

- Increased wear and corrosion resistance,
- Improved surface finish and appearance,
- Enhanced surface uniformity and purity, and
- Sealing of surface cracks and pores.

Even though it has long been known that these benefits can be obtained by RMRS, the process has not been used for large-scale commercial applications because of key technological barriers: the inability to produce broad area, efficient and intense energy sources capable of treating large surface areas, and the inability to deliver consistent treatments to complex and non-uniform surfaces.

Recently, these barriers to large scale RMRS have been overcome and the process has been implemented by QM Technologies in Albuquerque, NM, with its commercialization of the Ion Beam Surface Treatment (IBEST™) process that was developed and patented by Sandia National Laboratories in Albuquerque, NM. This process uses a rapidly pulsed (1–2 Hz), broad area (25–50 square cm) ion beam to melt and re-

solidify the top several microns (up to .0005") of the surfaces of most materials. The process causes the surface to be molten for about a microsecond, and the surface cools by rapid thermal diffusion at approximately a billion degrees a second. Because the energy is rapidly injected into only the top few microns of the surface, the average temperature of parts does not typically exceed 100 degrees centigrade. This latter fact allows tight dimensional tolerances (less than a few ten thousandths of an inch) to be maintained.

Examples of the effects of the RMRS process on typical gears and gear materials are presented below.

Creation of a Uniform Microstructure and Removal of Defects Using RMRS

Figure 1 shows the effect of RMRS on the microstructure of a typical carburized, 8000 series gear steel. This figure shows cross sections of the near surface of the steel before and after treatment. The samples were prepared by polishing and etching to reveal the microstructure of the materials.

The treated surface region in the cross section micrograph is essentially featureless. Detailed studies of pulsed ion beam treated steel have shown that the grain size of material in the rapidly re-solidified region is typically less than 100 nm. The process creates a homogeneous, equiaxed grain structure. In addition, melting followed by rapid resolidification results in



Figure 1—Metallographic cross section of untreated and RMRS processed, carburized, 8000 series gear material. The RMRS processed region is featureless, which is a result of its very small grain size and homogeneous, equiaxed structure.

a uniform redistribution of material in the region and removes localized defects like grain boundary oxidation, grain boundary carbide precipitates, and inclusions.

In carburized steels, rapid melting and re-solidification also creates a near surface region with retained austenite contents in the range of 20% to 30%. It has been reported in the literature that high levels of retained austenite can be beneficial in low cycle fatigue environments in which the enhanced strain at crack tips can cause an austenite to martensite transformation that helps to arrest further crack growth. It has also been reported that retained austenite with very small grain size can enhance high cycle fatigue life. The micrograph in Figure 1 shows that the grain size of the retained austenite in an RMRS transformed layer is very small and unresolvable by standard optical microscopy.

Further modification of the microstructure of the RMRS region has been achieved by subsequent thermal or mechanical processing. Heat treatment, cryogenic treatment, and shot peening have been shown to adjust the level of retained austenite in the transformed region. These post-RMRS processing options allow for customization of the microstructure to fit the requirements of different application environments.

In general, the RMRS process, when applied to metals, will result in the creation of a near surface region with very small, equiaxed grains and a uniform distribution of alloying elements. Defects composed of high vapor pressure constituents (for instance, inclusions) are often removed by the very rapid thermal excursion provided by RMRS.

In addition, features such as cracks and pores may be sealed by the melting process. For example, it has been demonstrated that there is sufficient time during RMRS for molten material to re-flow and fill pores and cracks that are up to several micrometers in size. This capability has facilitated the use of the RMRS process for sealing near surface porosity in metal injection molded parts and higher density (density > 94%) conventional powder metal parts.

Creation of a Smoother,

Isotropic Topography Using RMRS

As mentioned above, rapid melting and re-solidification results in the short range flow of material. As a result, the roughness and topography of a machined surface will be changed by RMRS.

For example, Figure 2 shows scanning electron microscope (SEM) images of the surface of the crown of a carburized steel gear tooth before and after RMRS. The untreated surface exhibits directional machine lines. In contrast, the surface of the treated gear shows that machine lines have been replaced by a longer wavelength, isotropic, shallow dimpled surface.

The treated gear surfaces were characterized using a Wyko NT-2000 optical profilometer. The root mean square roughness (Rq) and skewness (Rsk) of the surface features were measured. The average Rq of this gear was reduced by RMRS from an average of 90 +/- 14 micro-inches to 55 +/- 6 microinches. Other work shows that RMRS can reduce the roughness of metal surfaces by up to a factor of two to three.



Directional Machine Lines on the Gear Crown

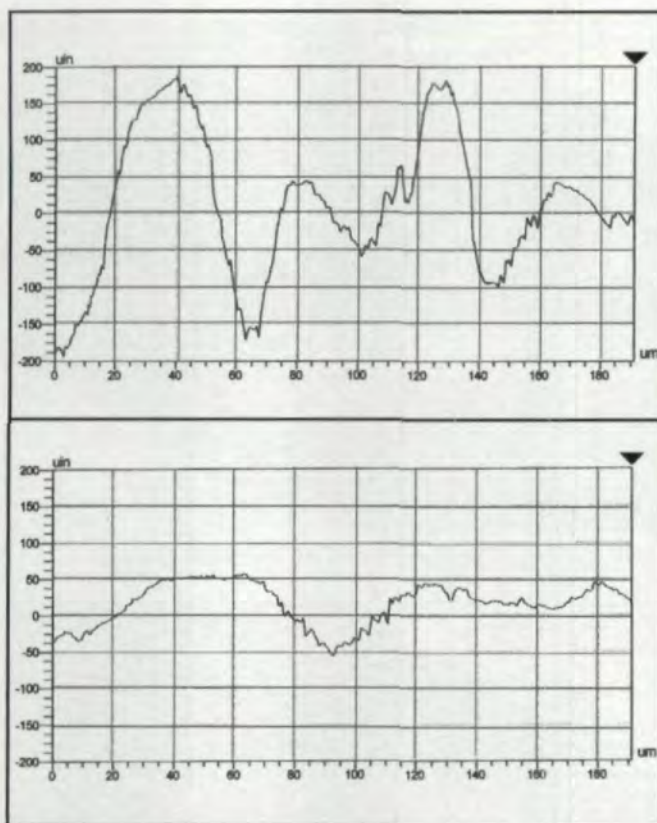


Isotropic Shallow "Valleys" on the Gear Crown

Figures 2a and 2b—Scanning electron microscope images (magnification = 250 X) of untreated and RMRS treated crowns of carburized steel gear teeth. The RMRS process transformed the directional, machined topography of the untreated surface to an isotropic, longer wavelength, and slightly dimpled topography.

A line profile taken from the surface of the untreated and RMRS treated surfaces of a carburized steel gear are shown in Figures 3a and 3b. These plots show that RMRS reduces the amplitude and increases the average wavelength of the surface features.

Finally, skewness is a measurement of the asymmetry of the surface profile about the mean surface position. A positive skewness indicates that the most distant lying points on a surface profile are proportionately above the mean surface while a negative skewness indicates the most distant lying points are proportionately below the mean surface. The skewness measurements of this gear showed that the average Rsk for the machined surface was approximately 0.0 +/- 0.2 and the average Rsk of a rapidly melted and re-solidified surface was approximately -0.3 +/- 0.2. This change in skewness reflects the creation of shallow "valleys" in the surface that is observed in Figure 2. The slightly negative skewness also suggests that the treated surface has few spikes or bumps that



Figures 3a and 3b—Surface profile scans of untreated and treated crowns of carburized, steel gear teeth taken using an optical profilometer. The RMRS process reduced the root-mean-square roughness from 90 micro-inches to 55 micro-inches and increased the wavelength of the characteristic surface features.

will typically wear away quickly in sliding wear environments. The presence of shallow valleys in the surface can also help with reducing run-in time while increasing lubrication retention in sliding applications. For example, work conducted by QMT on the RMRS processing of steel piston surfaces has shown that this type of surface significantly reduces friction in sliding applications.


The changes in the surface topography caused by RMRS processing have also been shown to significantly improve the adhesion of very hard coatings. In addition to removing contamination from the surface, the RMRS process reduces the population of stress enhancing, de-cohesion points at the hard coating-substrate interface. The increase in surface cleanliness and homogeneity combined with the creation of a smoother, isotropic finish by RMRS creates a better surface for coating.

Conclusions

When applied to gears, the RMRS process has been shown to:

1. Transform the microstructure of the gear material into a very fine grained, equiaxed and homogeneous microstructure with a reduced population of cracks and pores, and
2. Create surfaces that have isotropic, shallow, long wavelength, dimpled features rather than directional machined surface features of shorter characteristic wavelength.

These changes can result in decreased wear rate and increased fatigue life of gear materials. In addition, QMT has shown that the IBEST rapid melting and re-solidification process can be used to significantly enhance the performance

of surfaces in many diverse applications. For example, IBEST processing has increased the lubricity and wear resistance of metal injection molded parts, increased the lifetime of steel flexible rotary dies by up to a factor of ten, and extended the lifetime of cemented carbide tools used to machine cast iron by two to three times. 

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Gibson Heat Treat Inc.
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Horizon Steel Treating, Inc.
Hudapack Metal Treating
—Addison
Hudapack Metal Treating
—Glendale Heights
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Ipsen International Inc.
—Atmosphere Products
Ironbound Heat Treating
Irwin Automation Inc.
Jasco Heat Treating Inc.
Kowalski Heat Treating Co.
Lindberg Heat Treating Co.
—Eden Prairie
Lindberg Heat Treating Co.
—Lansing
Lindberg Heat Treating Co.
—Melrose Park
Lindberg Heat Treating Co.
—Rancho Dominguez
Lindberg Heat Treating Co.
—Rosemont
Lindberg Heat Treating Co.
—Tulsa
M & M Heat Treat
Metal Improvement Co. Inc.
—Wichita
Metal Treating Inc.
Metals Technology Corp.
Metlab
Metro Steel Treating
Met-Tek Inc.—Clackamas
Midwestern Machinery Co.
National Metal Processing
Paulo Products Co.
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Paulo Products Co.
—Murfreesboro
Paulo Products Co.
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Sonee Heat Treating Corp.
Southeastern Heat Treating
Steel Treating, Inc.
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—Orlando

Suncoast Heat Treat, Inc.
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Treating Div.
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Advanced Heat Treat Corp.
—Waterloo/Burton
Advanced Heat Treat Corp.
—Waterloo/Midport
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Alco Heat Treating Corp.
Alpha Heat Treating
Benedict-Miller Inc.
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Brazing Co.
Central Kentucky
Processing, Inc.
Clearing Niagara
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Gibson Heat Treat Inc.
Hauni Richmond Inc.
Hi-Tech Steel Treating Inc.
Horizon Steel Treating, Inc.
Impact Strategies, Inc.
Ipsen International Inc.
—Atmosphere Products
Ironbound Heat Treating
Jasco Heat Treating Inc.
Metal Improvement Co., Inc.
—Emigsville
Metlab
Midwestern Machinery Co.
Paulo Products Co.
—Kansas City
Paulo Products Co.
—Memphis
Paulo Products Co.
—St. Louis
Peters' Heat Treating Inc.
Rochester Steel Treating
Works, Inc.
Speciality Heat Treating
—Elkhart
Speciality Heat Treating
—Grand Rapids
Speciality Heat Treating
—Holland
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Thermal Metal Treating Inc.
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Alfe Heat Treating
Alfred Heller Heat Treating
Alliance Metal Treating
Alpha Heat Treating
American Heat Treating
American Metal Treating Inc.
Beehive Heat Treating
Benedict-Miller Inc.
Bennett Heat Treating &
Brazing Co.
Bodycote Thermal
Processing—Canton
Bodycote Thermal
Processing—Ft. Worth
Bodycote Thermal
Processing—Tulsa
Burbank Steel Treating, Inc.

Century Sun Metal Treating
Cincinnati Flame
Hardening Co.
Cincinnati Steel Treating Co.
Delphi Engineering
Detroit Steel Treating Co.
Dixie Heat Treating
Dreher Heat Treating
East-Lind Heat Treat
Edwards Heat Treating
Service
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Engineered Heat Treat, Inc.
Erie Steel Treating Inc.
Euclid Heat Treating
Fenton Heat Treating
Flame Metals Processing
FPM Ipsen Heat Treating
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General Metal Heat
Treating Inc.
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Heat Treating Co., Inc.
H & S Heat Treating
Hanson-Balk Steel Treating
Hauni Richmond Inc.
Hi-Tech Steel Treating Inc.
Hi-TecMetal Group
Corporate HQ
Horizon Steel Treating, Inc.
HTG Aerobraz
HTG IMT-York
Hudapack Metal Treating
—Elkhorn
Industrial Metal Treating
Industrial Steel Treating Inc.
Jasco Heat Treating Inc.
JCS Engineering &
Development Corp.
Kowalski Heat Treating Co.
Lindberg Heat Treating Co.
—Eden Prairie
Lindberg Heat Treating Co.
—Melrose Park
Lindberg Heat Treating Co.
—Rancho Dominguez
Lindberg Heat Treating Co.
—Rochester
Lindberg Heat Treating Co.
—Rosemont
Lindberg Heat Treating Co.
—Solon
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—Tulsa
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Merit Gear Heat Treating
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—Columbus
Metal Improvement Co., Inc.
—Milwaukee
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Metlab
Met-Tek Inc.—Clackamas
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Midwest Flame Hardening
Midwestern Machinery Co.
Modern Industries, Inc.
Modern Steel Treating
Nettleton Steel Treating
Oakland Metal Treating Co.
Paulo Products Co.
—Nashville
Paulo Products Co.
—St. Louis
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Phoenix Heat Treating Inc.
Progressive Steel Treating
Racine Heat Treating Co.
Speciality Heat Treating
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—Holland
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—Farmington Hills
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Tractech Inc.
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Universal Heat Treating
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Washington Metallurgical
Services
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—Lansing
Lindberg Heat Treating Co.
—Melrose Park
Lindberg Heat Treating Co.
—Rochester
Lindberg Heat Treating Co.
—Rosemont
Metlab

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Processing—Canton
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Processing—Ft. Worth
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—Cleveland
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Treating Corp.
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Hinderlter Heat Treating Inc.
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Hi-TecMetal Group
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 Alfred Heller Heat Treating
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 Works, Inc.
 Rotation Products Corp.
 S.K.S. Heat Treating Inc.
 Shore Metal Technology
 Solar Atmospheres Inc.
 Sonee Heat Treating Corp.
 Southeastern Heat Treating
 Speciality Heat Treating
 —Elkhart
 Speciality Heat Treating
 —Grand Rapids
 Speciality Heat Treating
 —Holland
 Speciality Heat Treating Inc.
 —Athens
 Speciality Steel Treating Inc.
 —Farmington Hills
 Speciality Steel Treating Inc.
 —Fraser
 Steel Treating, Inc.
 Steel Treating
 Sun Steel Treating Inc.
 Suncoast Heat Treat, Inc.
 —Palm Beach Gardens
 Suncoast Heat Treating
 —Pompano Beach
 Suncoast Heat Treating
 —Tampa
 Superior Metal Treating
 Syracuse Heat Treating
 T. N. Woodworth Inc.
 Thermal Brazing Inc.
 Thermal Metal Treating
 Thermet Inc.
 Thermo Electron Metal Treating Div.
 Therm-Tech of Waukesha
 Tractech Inc.
 Treat All Metals
 Trutec Industries
 Universal Heat Treating
 Washington Metallurgical
 Services
 Westside Flame Hardening

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 Beehive Heat Treating Inc.
 Benedict-Miller Inc.
 Bennett Heat Treating & Brazing Co.
 Bodycote Thermal Processing—Canton
 Burbank Steel Treating Inc.
 Caterpillar Industrial Products Inc.
 Century Sun Metal

Treating
 Cincinnati Gear Co.
 Cincinnati Steel Treating
 Detroit Steel Treating Co.
 Edwards Heat Treating Service
Engineered Heat Treat, Inc.
 Euclid Heat Treating
 Fairfield Mfg. Co.
 Fenton Heat Treating
 Fox Steel Treating Co.
 Franklin Steel Treating
 Gibson Heat Treat Inc.
Gleason Works
 Grand Rapids Commercial Heat Treating Co., Inc.
 Heat Treat Corp. of America
 Hi-TecMetal Group
Corporate HQ
 Hy-Vac Technologies Inc.
 Illiana Heat Treating Inc.
 Industrial Steel Treating
 Ironbound Heat Treating
 John V. Potero Co.
 Kowalski Heat Treating Co.
 Lindberg Heat Treating Co.
 —Racine
 Lindberg Heat Treating Co.
 —Rosemont
 Metallurgical Inc.
Metlab
 Oakland Metal Treating
 Ohio Metallurgical Service
 Paulo Products Co.
 —Bessemer
 Paulo Products Co.
 —Kansas City
 Paulo Products Co.
 —Memphis
 Paulo Products Co.
 —Nashville
 Phoenix Heat Treating Inc.
 Roboduction Thermal Processing
 Rotation Products Corp.
 Speciality Steel Treating Inc.
 —Farmington Hills
 Speciality Steel Treating Inc.
 —Fraser
 Sun Steel Treating Inc.
 Suncoast Heat Treat, Inc.
 —Palm Beach Gardens
 Therm Tech of Waukesha
 Treat All Metals
 Walker Heat Treating
 Washington Metallurgical
 Services
Westside Flame Hardening

FLAME HARDENING

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 American Heat Treating
 Beehive Heat Treating Inc.
 Bennett Heat Treating & Brazing Co.
 Bodycote Thermal Processing—Canton
 Bodycote Thermal Processing—Ft. Worth
 Bodycote Thermal Processing—Tulsa
 Bowdill Co.
 California Surface Hardening, Inc.
 Calumet Surface Hardening
 Certified Heat Treating Inc.
 Chicago Flame Hardening
 Chicago Induction Metal Treating Corp.
 Cincinnati Flame Hardening Co.
 Cincinnati Gear Co.
 Cincinnati Steel Treating
 Clearing Niagara
 Cleveland Flame Hardening

Coleman Commercial Heat Treating
 Detroit Flame Hardening
 Detroit Steel Treating Co.
 Drever Heat Treating
 East-Lind Heat Treat
 Economy Flame Hardening, Inc.
 Edwards Heat Treating Serv.
 Erie Steel Treating Inc.
 Flame Hardening Co. of California
 Fox Steel Treating Co.
 Franklin Steel Treating
 Gibson Heat Treat Inc.
 Good Earth Tools
 Grand Rapids Commercial Heat Treating Co., Inc.
 Hansen-Balk Steel Treating
 Hauni Richmond Inc.
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 HTG IMF-York
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 Lindberg Heat Treating Co.—Dallas
 Lindberg Heat Treating Co.—Racine
 Lindberg Heat Treating Co.—Rosemont
 Lindberg Heat Treating Co.—St. Louis
 M & M Heat Treat
 Metal Improvement Co. Inc.—Wichita
 Metal Improvement Co., Inc.—Columbus
 Metal Treating Inc.
 Metallurgical Inc.
 Metlab
 Met-Tek Inc.—Clackamas
 Met-Tek Inc.—Racine
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 Mid-West Flame Hardening
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 Oregon Induction Corp.
 Penna Flame Industries Inc.
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 Speciality Heat Treating Inc.—Athens
 Steel Treaters, Inc.
 Suncoast Heat Treat, Inc.—Orlando
 Suncoast Heat Treat, Inc.—Palm Beach Gardens
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 Edwards Heat Treating Service
 Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
 Euclid Heat Treating
 Fairfield Mfg. Co.
 Fenton Heat Treating
 FPM Heat Treating—Elk Grove
 FPM Milwaukee
 Gibson Heat Treat Inc.
 Hanson-Balk Steel Treating
 Heat Treat Corp. of America
 Heat-Treating Inc.
 Hinderliter Heat Treating Inc.—Anaheim
 Hinderliter Heat Treating Inc.—Tarzana
 Hi-Tech Metallurgical Co.—Huron
 Illiana Heat Treating Inc.
 Induction Metal Treating Co.
 Industrial Metal Treating
 Industrial Steel Treating Inc.
 Ipsen International Inc.—Atmosphere Products
 Kowalski Heat Treating Co.
 Lake County Steel Treating
 Lawrence Industries, Inc.
 Lindberg Heat Treating Co.—Berlin
 Lindberg Heat Treating Co.—Dallas
 Lindberg Heat Treating Co.—Eden Prairie
 Lindberg Heat Treating Co.—Houston
 Lindberg Heat Treating Co.—Lansing
 Lindberg Heat Treating Co.—Melrose Park
 Lindberg Heat Treating Co.—Racine
 Lindberg Heat Treating Co.—Rancho Dominguez
 Lindberg Heat Treating Co.—Rosemont
 Lindberg Heat Treating Co.—Solon
 Lindberg Heat Treating Co.—St. Louis
 Lindberg Heat Treating Co.—Tulsa
 Lindberg Heat Treating Co.—Worcester
 Merit Gear Heat Treating
 Metal Treating Inc.
 Metallurgical Inc.
 Metals Technology Corp.
 Metlab

Met-Tek Inc.—Racine
 Mid-West Flame Hardening
National Broach & Machine
 National Induction Heating
 Nettleton Steel Treating
 Ohio Metallurgical Service
 P & L Heat Treating & Grinding
 Paulo Products Co.—St. Louis
 Phoenix Heat Treating Inc.
 Pitt-Tex Inc.
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 Rotation Products Corp.
 S.K.S. Heat Treating Inc.
 Scot Forge
 Sonee Heat Treating Corp.
 Speciality Steel Treating Inc.—Farmington Hills
 Speciality Steel Treating Inc.—Fraser
 State Heat Treat Inc.
 Suncoast Heat Treating—Pompano Beach
 Superior Metal Treating
 Syracuse Heat Treating
 T. N. Woodworth Inc.
 Thermo Electron Metal Treating Div.
 Treat All Metals
 Wohler Corp.

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 Alpha Heat Treaters
 American Heat Treating
 Atmosphere Annealing Inc.
 Atmosphere Furnace Co.
 Beehive Heat Treating Inc.
 Bennett Heat Treating & Brazing Co.
 Bodycote Thermal Processing—Canton
 Bodycote Thermal Processing—Ft. Worth
 Bodycote Thermal Processing—Tulsa
 Bomak Corp.
 Burbank Steel Treating Inc.
 Carolina Commercial Heat Treating
 Century Sun Metal Treating
 Certified Heat Treating Inc.
 Certified Metal Craft, Inc.
 Cincinnati Gear Co.
 Cincinnati Steel Treating Co.
 City Steel Treating Inc.
 Clearing Niagara
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 Custom Heat Treating Co.
 Detroit Steel Treating Co.
 Disston Precision Inc.
 Dixie Machine & Heat Treating Inc.
 Drever Heat Treating
 East Carolina Metal Treating Inc.
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 Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
 Erie Steel Treating Inc.
 Euclid Heat Treating
 Fairfield Mfg. Co.
 Federal Machine Co.
 Feinblanking Ltd.
 Fenton Heat Treating

Flame Metals Processing
 Fox Steel Treating Co.
 FPM Heat Treating—Elk Grove
 FPM Ipsen Heat Treating
 FPM Milwaukee
 Franklin Steel Treating
 General Heat Treating Corp.
 General Metal Heat Treating Inc.
 Gibson Heat Treat Inc.
 Hansen-Balk Steel Treating
 Hauni Richmond Inc.
 Heat Treat Corp. of America
 Heat-Treating Inc.
 Hinderliter Heat Treating Inc.—Tulsa
 Hi-Tech Steel Treating Inc.
 Hi-TecMetal Group
 Corporate HQ
 Horoburgh & Scott
 Hudapack Metal Treating—Elkhorn
 Huron
 Illiana Heat Treating Inc.
 Induction Metal Treating Co.
 Industrial Metal Treating
 Industrial Steel Treating Inc.
 Ipsen International Inc.—Atmosphere Products
 Ipsen International Inc.—Vacuum Products
 Irwin Automation Inc.
 Jasco Heat Treating Inc.
 John V. Potero Co.
 Kowalski Heat Treating Co.
 Lindberg Heat Treating Co.—Lansing
 Lindberg Heat Treating Co.—Melrose Park
 Lindberg Heat Treating Co.—Rosemont
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 Merit Gear Heat Treating
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 Metallurgical Processing Inc.
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 Metals Technology Corp.
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 Met-Tek Inc.—Clackamas
 Met-Tek Inc.—Racine
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 Paulo Products Co.—St. Louis
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 Southeastern Heat Treating
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Hydro-Vac
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—Vacuum Products
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Lindberg Heat Treating Co.
—Rosemont
Lindberg Heat Treating Co.
—Sturtevant
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Modern Industries, Inc.
MPT America
National Broach & Machine
Progressive Engineering
Solar Atmospheres Inc.
Sonec Heat Treating Corp.
Sun Steel Treating Inc.

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Laser Machining, Inc.
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—St. Laurent
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 —Elk Grove
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 —Tulsa
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 —Solon
 Lindberg Heat Treating Co.
 —St. Louis
 Lindberg Heat Treating Co.
 —Tulsa
 Lindberg Heat Treating Co.
 —Worcester
 Metals Technology Corp.
 Metal-Tec Heat Treating

Metlab
 Metro Steel Treating
Modern Industries, Inc.
 Nitrex Metal Technologies
 —Burlington
 Nitrex Metal Technologies I
 —St. Laurent
 Nitrotec Surface
 Engineering—Michigan
 Nitrotec Surface
 Engineering—Ohio
 Paulo Products Co.
 —Bessemer
 Paulo Products Co.
 —Kansas City
 Paulo Products Co.
 —Memphis
 Paulo Products Co.
 —Nashville
 Paulo Products Co.
 —St. Louis
 Peters' Heat Treating Inc.
 Phoenix Heat Treating Inc.
 Rochester Steel Treating
 Works, Inc.
 Rotation Products Corp.
 Shore Metal Technology
 Speciality Heat Treating
 —Elkhart
 Speciality Heat Treating
 —Grand Rapids
 Speciality Heat Treating
 —Holland
 Speciality Heat Treating Inc.
 —Athens
 Suncoast Heat Treating
 —Pompano Beach
 Syracuse Heat Treating
 Therm Tech of Waukesha
 Treat All Metals
 Tratec Industries

NORMALIZING

A.F.C.-Pifco
 Abbott Furnace Company
 ABS Metallurgical
 Processors
 Accurate Steel Treating
 Advanced Heat Treat Corp.
 —Waterloo/Burton
 Advanced Metallurgical
 Technology, Inc.
 Advanced Thermal
 Technologies, Inc.
Ajax Magnethermic Corp.
 Albany Metal Treating
 Alco Heat Treating Corp.
 Alfe Heat Treating
 Alfred Heller Heat Treating
 Alliance Metal Treating
 Alpha Heat Treating
 American Heat Treating
 American Metal Treating Inc.
 Atmosphere Annealing Inc.
 Atmosphere Furnace Co.
 Beehive Heat Treating Inc.
 Benedict-Miller Inc.
 Bennett Heat Treating &
 Brazing Co.
 Bodycote Thermal
 Processing—Canton
 Bodycote Thermal
 Processing—Ft. Worth
 Bodycote Thermal
 Processing—Tulsa
 Bomak Corp.
 Braddock Metallurgical
 Braddock Metallurgical
 —Alabama
 Brazing & Metal Treating
 —Cleveland
 Brazing & Metal Treating
 —Kentucky
 Brazing & Metal Treating
 —Minnesota
 Brite Brazing
 Bowdill Co.

Burbank Steel Treating
 Cal-Doran Division
 Carolina Commercial Heat
 Treating
 Caterpillar Industrial
 Products
 Century Sun Metal Treating
 Certified Heat Treating Inc.
 Cincinnati Flame
 Hardening Co.
 Cincinnati Gear Co.
 Cincinnati Steel Treating
 City Steel Treating Inc.
 Clearing Niagara
 Coleman Commercial
 Heat Treating
 Commercial Steel Treating
 Custom Heat Treating Co.
 Delavan Steel Treating
 Delphi Engineering
 Detroit Steel Treating Co.
 Diamond Heat Treating
 Disston Precision Inc.
 Dixie Heat Treating Co.
 Dixie Machine & Heat
 Treating Inc.
 Drever Heat Treating
 Dynamic Metal Treating
 East Carolina Metal
 Treating Inc.
 East-Lind Heat Treat
 Eckel Heat Treat
 Edwards Heat Treating
 Service
 Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
 Erie Steel Treating Inc.
 Euclid Heat Treating
 Fairfield Mfg. Co.
 Federal Machine Co.
 Feinblanking Ltd.
 Fenton Heat Treating
 Flame Metals Processing
 Fox Steel Treating Co.
 FPM Heat Treating
 —Elk Grove
 FPM Ipsen Heat Treating
 FPM Milwaukee
 Franklin Steel Treating
 Gear Company of America
 General Heat Treating
 General Metal Heat
 Treating Inc.
 Geo. H. Porter Steel
 Treating Co.
 Gibson Heat Treat Inc.
 Grand Rapids Commercial
 Heat Treating Co., Inc.
 H & S Heat Treating
 Hansen-Balk Steel
 Treating
 Hauni Richmond Inc.
 Heat Treat Corp. of
 America
 Heat Treating Services
 Heat-Treating Inc.
 Hinderliter Heat Treating Inc.
 —Anaheim
 Hinderliter Heat Treating Inc.
 —Dallas
 Hinderliter Heat Treating Inc.
 —Tazana
 Hinderliter Heat Treating Inc.
 —Tulsa
 Hi-Tech Aero
 Hi-Tech Steel Treating Inc.
 Hi-TecMetal Group
 Corporate HQ
 Horizon Steel Treating, Inc.
 Horsburgh & Scott Co.
 HTG Aerobrazing
 HTG IMT-Duncan
 HTG IMT-York
 HTG Metal Methods
 Hudapack Metal Treating
 —Elkhorn
 Hudapack Metal Treating
 —Glendale Heights
 Huron
 Hy-Vac Technologies Inc.

Illiana Heat Treating Inc.
 Induction Metal Treating
 Industrial Metal Treating
 Industrial Steel Treating Inc.
 Ipsen International Inc.
 —Atmosphere Products
 Ipsen International Inc.
 —Vacuum Products
 Ironbound Heat Treating
 Irwin Automation Inc.
 Jasco Heat Treating Inc.
 John V. Potero Co.
 Kowalski Heat Treating Co.
 Lake County Steel Treating
 Lawrence Industries, Inc.
 Lindberg Heat Treating Co.
 —Berlin
 Lindberg Heat Treating Co.
 —Dallas
 Lindberg Heat Treating Co.
 —Eden Prairie
 Lindberg Heat Treating Co.
 —Houston
 Lindberg Heat Treating Co.
 —Melrose Park
 Lindberg Heat Treating Co.
 —Racine
 Lindberg Heat Treating Co.
 —Rancho Dominguez
 Lindberg Heat Treating Co.
 —Rosemont
 Lindberg Heat Treating Co.
 —Solon
 Lindberg Heat Treating Co.
 —St. Louis
 Lindberg Heat Treating Co.
 —Tulsa
 Lindberg Heat Treating Co.
 —Worcester
 M & M Heat Treat
 Magnum Metal Treating
 Master Heat Treating Inc.
 Merit Gear Heat Treating
 Metal Improvement Co., Inc.
 —Wichita
 Metal Improvement Co., Inc.
 —Columbus
 Metal Improvement Co., Inc.
 —Emigsville
 Metal Treating Inc.
 Metal Treating & Research
 Metal Treating Inc.
 Metallurgical Inc.
 Metallurgical Processing
 Metals Engineering Inc.
 Metals Technology Corp.
 Metlab
 Metro Steel Treating
 Met-Tek Inc.—Clackamas
 Met-Tek Inc.—Racine
 Midland Metal Treating
Modern Industries, Inc.
 Modern Steel Treating
 Mountain Metallurgical
National Branch & Machine
 National Metal Processing,
 Nettleton Steel Treating
 Nitro-Vac Heat Treating
 Oakland Metal Treating
 Ohio Metallurgical Service
 Oregon Induction Corp.
 P & L Heat Treating &
 Grinding
 Partek Laboratories Inc.
 Paulo Products Co.
 —Bessemer
 Paulo Products Co.
 —Kansas City
 Paulo Products Co.
 —Memphis
 Paulo Products Co.
 —Nashville
 Paulo Products Co.
 —St. Louis
 Pennsylvania Metallurgical
 Peters' Heat Treating Inc.
 Phoenix Heat Treating Inc.
 Pitt-Tex Inc.
 Precision Heat Treating
 Progressive Steel Treating
 Racine Heat Treating Co.

Rochester Steel Treating
 Works, Inc.
 Rotation Products Corp.
 S.K.S. Heat Treating Inc.
 Scot Forge
 Shore Metal Technology
 Sonee Heat Treating Corp.
 Southeastern Heat Treating
 Speciality Heat Treating
 —Elkhart
 Speciality Heat Treating
 —Grand Rapids
 Speciality Heat Treating
 —Holland
 Speciality Heat Treating Inc.
 —Athens
 Speciality Steel Treating Inc.
 —Farmington Hills
 Speciality Steel Treating Inc.
 —Fraser
 State Heat Treat Inc.
 Steel Treating, Inc.
 Sun Steel Treating Inc.
 Suncoast Heat Treat, Inc.
 —Orlando
 Suncoast Heat Treat, Inc.
 —Palm Beach Gardens
 Suncoast Heat Treating
 —Pompano Beach
 Suncoast Heat Treating
 —Tampa
 Superior Metal Treating
 Syracuse Heat Treating
 T. N. Woodworth Inc.
 Thermal Metal Treating Inc.
 Thermo Electron Metal
 Treating Div.
 Thermo Treating Ltd.
 Thermo-Tech of Waukesha
 Tratech Inc.
 Treat All Metals
 Trojan Heat Treat Inc.
 Universal Heat Treating
 Vacu Braze
 Walker Heat Treating
 Washington Metallurgical
 Services
 Weiss Industries Inc.
 Westside Flame Hardening
 Wohler Corp.

PLASMA CARBURIZING

Cincinnati Gear Co.
 Gleason Pfauter Hurth
 Cutting Tools
 Ionex, Inc.
 Ipsen International Inc.
 —Vacuum Products

PRESS QUENCHING

A.F.C.-Pifco
 Alfred Heller Heat Treating
 Atmosphere Furnace Co.
 Benedict-Miller Inc.
 Bennett Heat Treating &
 Brazing Co.
 Caterpillar Industrial
 Products Inc.
 Cincinnati Gear Co.
 Cincinnati Steel Treating
Engineered Heat Treat, Inc.
 Euclid Heat Treating
 Fairfield Mfg. Co.
 Franklin Steel Treating
 Gibson Heat Treat Inc.
Gleason Works.
 Heat Treat Corp. of
 America
 Hi-Tech Metallurgical Co.
 Impact Strategies, Inc.
 Industrial Steel Treating
 Kowalski Heat Treating Co.
 Lindberg Heat Treating Co.

—Racine
 Lindberg Heat Treating Co.
 —Rosemont
 Metallurgical Inc.
 Metlab
 Mountain Metallurgical
 Ohio Metallurgical Service
 Paulo Products Co.
 —St. Louis
 Phoenix Heat Treating Inc.
 Roboduction Thermal
 Processing
 Speciality Steel Treating Inc.
 —Farmington Hills
 Speciality Steel Treating Inc.
 —Fraser
 Tratech Inc.
 Treat All Metals
 Washington Metallurgical
 Services

SALT BATH NITRIDING

Bodycote Thermal
 Processing—Canton
 Bomak Corp.
 Cal-Doran Division
 Century Sun Metal
 Treating
 Cincinnati Gear Co.
 Commercial Steel Treating
 East-Lind Heat Treat
Engineered Heat Treat, Inc.
 Flame Metals Processing
 Fox Steel Treating Co.
 Franklin Steel Treating
 H & M Metal Processing
 HI TecMetal Group
 —Cleveland
 Illiana Heat Treating Inc.
 Induction Metal Treating
 Lake County Steel Treating
 Metal Treating Inc.
 Met-Tek Inc.—Clackamas
 Nitro-Vac Heat Treating
 O & W Heat Treat Inc.
 Tratec Industries
 Wear-Ever Surface
 Treating Corp.
 Weiss Industries Inc.
 Westside Flame Hardening

SHOT PEENING

Alpha Heat Treating
 Bodycote Thermal
 Processing—Canton
 Coleman Commercial Heat
 Treating
 Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
 Flame Metals Processing
 Gibson Heat Treat Inc.
 Horizon Steel Treating
 Jasco Heat Treating Inc.
 Kowalski Heat Treating Co.
 Lindberg Heat Treating Co.
 —Melrose Park
 Metal Improvement Co., Inc.
 —Addison
 Metal Improvement Co., Inc.
 —Blue Ash
 Metal Improvement Co., Inc.
 —Brampton
 Metal Improvement Co., Inc.
 —Carlstadt
 Metal Improvement Co., Inc.
 —Charlotte
 Metal Improvement Co., Inc.
 —Columbus
 Metal Improvement Co., Inc.
 —Grand Prairie
 Metal Improvement Co., Inc.
 —Houston

Metal Improvement Co., Inc.
—Lynwood
Metal Improvement Co., Inc.
—Maple Grove
Metal Improvement Co., Inc.
—Miami
Metal Improvement Co., Inc.
—Milwaukee
Metal Improvement Co., Inc.
—Phoenix
Metal Improvement Co., Inc.
—Romulus
Metal Improvement Co., Inc.
—Santa Ana
Metal Improvement Co., Inc.
—Twinsburg
Metal Improvement Co., Inc.
—Vernon
Metal Improvement Co., Inc.
—Wakefield
Metal Improvement Co., Inc.
—Wellington
Metal Improvement Co., Inc.
—West Babylon
Metal Improvement Co., Inc.
—Windsor
Paulo Products Co.
—Memphis
Superior Metal Treating
Syracuse Heat Treating
Tractech Inc.
Westside Flame Hardening

SINTERING

A.F.C.-Pifco
Abbott Furnace Company
Allread Products
Atmosphere Furnace Co.
Bennett Heat Treating &
Brazing Co.
Bodycote Thermal
Processing—Canton
Certified Metal Craft, Inc.
Cincinnati Gear Co.
Fluxtrol Manufacturing Inc.
Hinderliter Heat Treating Inc.
—Tarzana
Hi-TecMetal Group
Corporate HQ
HTG Aerobraz
Hydro-Vac
Hy-Vac Technologies Inc.
Illiana Heat Treating Inc.
Induction Metal Treating
Ionex, Inc.
Ipsen International Inc.
—Atmosphere Products
Ipsen International Inc.
—Vacuum Products
Metals Technology Corp.
Paulo Products Co.
—St. Louis
Pennsylvania Metallurgical
Progressive Steel Treating
Specialty Heat Treating Inc.
—Athens
Suncoast Heat Treating
—Pompano Beach

STEAM TREATING

A.F.C.-Pifco
Abbott Furnace Company
Accurate Ion Technologies
Advanced Heat Treating
Atmosphere Furnace Co.
Gleason Pfauter Hurth
Cutting Tools
Hi-Tech Steel Treating Inc.
Hi-TecMetal Group
Corporate HQ
Industrial Metal Treating
Ipsen International Inc.
—Atmosphere Products
Modern Industries, Inc.

Nettleton Steel Treating
Solar Atmospheres Inc.
Sun Steel Treating Inc.
Syracuse Heat Treating

STRAIGHTENING

ABS Metallurgical
Processors
Accurate Steel Treating
Alfe Heat Treating
Alfred Heller Heat Treating
American Metal Treating Inc.
Bennett Heat Treating &
Brazing Co.
Bodycote Thermal
Processing—Canton
Bodycote Thermal
Processing—Ft. Worth
Bodycote Thermal
Processing—Tulsa
Calumet Surface Hardening
Century Sun Metal Treating
Certified Metal Craft, Inc.
Chicago Flame Hardening
Cincinnati Flame
Hardening Co.
Cincinnati Steel Treating
Clearing Niagara
Commercial Induction
Detroit Flame Hardening
Detroit Steel Treating Co.
Dynamic Metal Treating
Economy Flame
Hardening, Inc.
Edwards Heat Treating
Service
Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
Erie Steel Treating Inc.
Euclid Heat Treating
Flame Metals Processing
Franklin Steel Treating
Gibson Heat Treat Inc.
Gleason Pfauter Hurth
Cutting Tools
Grand Rapids Commercial
Heat Treating Co., Inc.
Hi-Tech Steel Treating Inc.
Hi-TecMetal Group
Corporate HQ
Horizon Steel Treating, Inc.
Horsburgh & Scott Co.
Houston Flame Hardening
Hudapack Metal Treating
—Elkhorn
Jasco Heat Treating Inc.
Kowalski Heat Treating Co.
Lindberg Heat Treating Co.
—Dallas
Lindberg Heat Treating Co.
—Eden Prairie
Lindberg Heat Treating Co.
—Houston
Lindberg Heat Treating Co.
—Melrose Park
Lindberg Heat Treating Co.
—Racine
Lindberg Heat Treating Co.
—Rancho Dominguez
Lindberg Heat Treating Co.
—Rosemont
Lindberg Heat Treating Co.
—Solon
Lindberg Heat Treating Co.
—St. Louis
Lindberg Heat Treating Co.
—Sturtevant
Lindberg Heat Treating Co.
—Tulsa
Lindberg Heat Treating Co.
—Wichita
Lindberg Heat Treating Co.
—Worcester
M & M Heat Treat
Merit Gear Heat Treating
Metal Improvement Co.
—Wichita

Metal Improvement Co.
—Columbus
Metal Treating Inc.
Metallurgical Inc.
Metals Technology Corp.
Metlab
Met-Tek Inc.—Clackamas
Midwest Flame Hardening
Modern Steel Treating
National Broach & Machine
Oakland Metal Treating Co.
Oregon Induction Corp.
Paulo Products Co.
—Nashville
Paulo Products Co.
—St. Louis
Phoenix Heat Treating Inc.
Progressive Steel Treating
Racine Heat Treating Co.
Specialty Heat Treating
—Elkhart
Specialty Heat Treating
—Grand Rapids
Specialty Heat Treating
—Holland
Specialty Steel Treating Inc.
—Farmington Hills
Specialty Steel Treating Inc.
—Fraser
Sun Steel Treating Inc.
Suncoast Heat Treat, Inc.
—Palm Beach Gardens
Superior Metal Treating
Syracuse Heat Treating
Thermet Inc.
Therm-Tech of Waukesha
Tractech Inc.
Treat All Metals
Washington Metallurgical
Services
Westside Flame Hardening

STRESS RELIEVING

Abbott Furnace Company
ABS Metallurgical
Processors
Accurate Steel Treating
Advanced Heat Treat Corp.
—Monroe
Advanced Heat Treat Corp.
—Waterloo/Burton
Advanced Metallurgical
Technology, Inc.
Advanced Thermal
Technologies, Inc.
Ajax Magnethermic Corp.
Albany Metal Treating
Alco Heat Treating Corp.
Alfe Heat Treating
Alfred Heller Heat Treating
Alliance Metal Treating
Alpha Heat Treating
American Brazing
American Heat Treating
American Metal Treating Inc.
AMT—Monroe, Inc.
Atmosphere Annealing Inc.
Atmosphere Furnace Co.
Beehive Heat Treating Inc.
Benedict-Miller Inc.
Bennett Heat Treating &
Brazing Co.
Bodycote Thermal
Processing—Canton
Bodycote Thermal
Processing—Ft. Worth
Bodycote Thermal
Processing—Tulsa
Bomak Corp.
Bonal Technologies, Inc.
Bowditch Co.
Braddock Metallurgical
Braddock Metallurgical
—Alabama
Brazing & Metal Treating
—Cleveland
Brazing & Metal
Treating—Kentucky
Brazing & Metal
Treating—Minnesota
Brite Brazing
Brite Metal Treating Inc.
Burbank Steel Treating Inc.
Cal-Doran Division
Carolina Commercial Heat
Treating
Central Kentucky
Processing, Inc.
Century Sun Metal Treating
Certified Heat Treating Inc.
Certified Metal Craft, Inc.
Chicago Flame Hardening
Cincinnati Flame
Hardening Co.
Cincinnati Gear Co.
Cincinnati Steel Treating
City Steel Treating Inc.
Clearing Niagara
Cleveland Flame Hardening
Coleman Commercial Heat
Treating
Commercial Induction
Commercial Steel Treating
Cooperheat Inc.
Custom Heat Treating Co.
Delavan Steel Treating
Delphi Engineering
Detroit Flame Hardening
Detroit Steel Treating Co.
Diamond Heat Treating Co.
Disston Precision Inc.
Dixie Heat Treating Co., Inc.
Dixie Machine & Heat
Treating Inc.
Drever Heat Treating
Dynamic Metal Treating
East Carolina Metal
Treating Inc.
East-Lind Heat Treat
Eckel Heat Treat
Edwards Heat Treating
Service
Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
Erie Steel Treating Inc.
Euclid Heat Treating
Fairfield Mfg. Co.
Feinblanking Ltd.
Fenton Heat Treating
Flame Metals Processing
Fluxtrol Manufacturing Inc.
Fox Steel Treating Co.
FPM Heat Treating
—Elk Grove
FPM Ipsen Heat Treating
FPM Milwaukee
Franklin Steel Treating
General Heat Treating Corp.
General Metal Heat
Treating Inc.
Geo. H. Porter Steel
Treating Co.
Gibson Heat Treat Inc.
Gleason Pfauter Hurth
Cutting Tools
Global Heat Inc.
Grand Rapids Commercial
Heat Treating Co., Inc.
H & M Metal Processing
H & S Heat Treating
Hansen-Balk Steel Treating
Hauni Richmond Inc.
Heat Treat Corp. of
America
Heat Treating Services
Heat-Treating Inc.
Hinderliter Heat Treating Inc.
—Anaheim
Hinderliter Heat Treating Inc.
—Dallas
Hinderliter Heat Treating Inc.
—Tarzana
Hinderliter Heat Treating Inc.
—Tulsa
Hi-Tech Aero
Hi-Tech Metallurgical Co.
Hi-Tech Steel Treating Inc.
Hi-TecMetal Group

Corporate HQ
Horizon Steel Treating, Inc.
Horsburgh & Scott Co.
Houston Flame Hardening
HTG Aerobraz
HTG Copper Brazing
Industries
HTG IMT-Duncan
HTG IMT-York
HTG Metal Methods
Hudapack Metal Treating
—Elkhorn
Hudapack Metal Treating
—Glendale Heights
Huron
Hydro-Vac
Hy-Vac Technologies Inc.
Illiana Heat Treating Inc.
Impact Strategies, Inc.
Induction Metal Treating
Industries Services Inc.
Inductoheat, Inc.
Industrial Metal Treating
Industrial Steel Treating Inc.
Ionex, Inc.
Ipsen International Inc.
—Atmosphere Products
Ipsen International Inc.
—Vacuum Products
Ironbound Heat Treating
Irwin Automation Inc.
Jasco Heat Treating Inc.
JCS Engineering &
Development Corp.
John V. Potero Co.
Kowalski Heat Treating Co.
Lake County Steel Treating
Lawrence Industries, Inc.
Lindberg Heat Treating Co.
—Berlin
Lindberg Heat Treating Co.
—Dallas
Lindberg Heat Treating Co.
—Eden Prairie
Lindberg Heat Treating Co.
—Houston
Lindberg Heat Treating Co.
—Lansing
Lindberg Heat Treating Co.
—Melrose Park
Lindberg Heat Treating Co.
—Mexico
Lindberg Heat Treating Co.
—Racine
Lindberg Heat Treating Co.
—Rancho Dominguez
Lindberg Heat Treating Co.
—Rochester
Lindberg Heat Treating Co.
—Rosemont
Lindberg Heat Treating Co.
—Solon
Lindberg Heat Treating Co.
—St. Louis
Lindberg Heat Treating Co.
—Sturtevant
Lindberg Heat Treating Co.
—Tulsa
Lindberg Heat Treating Co.
—Worcester
M & M Heat Treat
Magnum Metal Treating
Mannings U.S.A.
Master Heat Treating Inc.
Merit Gear Heat Treating
Metal Improvement Co., Inc.
—Wichita
Metal Improvement Co., Inc.
—Columbus
Metal Improvement Co., Inc.
—Emigsville
Metal Treating Inc.
Metal Treating & Research
Metal Treating Inc.
Metallurgical Inc.
Metals Engineering Inc.
Metals Technology Corp.
Metal-Tec Heat Treating Inc.
Metlab
Metro Steel Treating
Met-Tek Inc.—Clackamas

Met-Tek Inc.—Racine
Michigan Flame
Hardening
Michigan Induction Inc.
Midland Metal Treating
Midwest Flame Hardening
Midwestern Machinery
Modern Industries, Inc.
Modern Metal Processing
Modern Steel Treating
Mountain Metallurgical
National Broach & Machine
National Induction Heating
National Metal Processing
Nettleton Steel Treating
Nitro-Vac Heat Treating
O & W Heat Treat Inc.
Oakland Metal Treating
Ohio Metallurgical Service
Oregon Induction Corp.
Partek Laboratories Inc.
Paulo Products Co.
—Bessemer
Paulo Products Co.
—Kansas City
Paulo Products Co.
—Memphis
Paulo Products Co.
—Nashville
Paulo Products Co.
—St. Louis
Penna Flame Industries
Pennsylvania Metallurgical
Peters' Heat Treating Inc.
Phoenix Heat Treating Inc.
Pitt-Tex Inc.
Precision Heat Treating
Progressive Steel Treating
Racine Heat Treating Co.
Roboduction Thermal
Processing
Rochester Steel Treating
Works, Inc.
Rotation Products Corp.
S.K.S. Heat Treating Inc.
Scot Forge
Shanafelt Mfg. Co.
Shore Metal Technology
Solar Atmospheres Inc.
Sonec Heat Treating Corp.
Southeastern Heat Treating
Specialty Heat Treating
—Elkhart
Specialty Heat Treating
—Grand Rapids
Specialty Heat Treating
—Holland
Specialty Steel Treating Inc.
—Athens
Specialty Steel Treating Inc.
—Farmington Hills
Specialty Steel Treating Inc.
—Fraser
State Heat Treat Inc.
Steel Treating, Inc.
Steel Treating
Sun Steel Treating Inc.
Suncoast Heat Treat, Inc.
—Orlando
Suncoast Heat Treat, Inc.
—Palm Beach Gardens
Suncoast Heat Treating
—Pompano Beach
Suncoast Heat Treating
—Tampa
Superior Metal Treating
Syracuse Heat Treating
T. N. Woodworth Inc.
Thermal Metal Treating
Thermo Electron Metal
Treating Div.
Thermo Treating Ltd.
Therm-Tech of Waukesha
Tocco, Inc.
Tractech Inc.
Treat All Metals
Trojan Heat Treat Inc.
Truete Industries
Universal Heat Treating
Vacu Braze
Washington Metallurgical

Services
Weiss Industries Inc.
Western Stress Inc.
Westside Flame Hardening

TEMPERING

A.F.C.-Pifco
Abbott Furnace Company
ABS Metallurgical
Processors
Accurate Steel Treating
Advanced Heat Treat Corp.
—Waterloo/Burton
Advanced Heat Treating
Advanced Metallurgical
Technology, Inc.
Advanced Thermal
Technologies, Inc.
Ajax Magnethermic Corp.
Albany Metal Treating
Alco Heat Treating Corp.
Alfe Heat Treating
Alfred Heller Heat Treating
Alliance Metal Treating
Alpha Heat Treating
American Brazing
American Cryogenics, Inc.
American Heat Treating
American Metal Processing
American Metal Treating Inc.
AP Westshore
Atmosphere Annealing Inc.
Atmosphere Furnace Co.
Beehive Heat Treating Inc.
Benedict-Miller Inc.
Bennett Heat Treating &
Brazing Co.
Bodycote Induction
Processing
Bodycote Thermal
Processing—Canton
Bodycote Thermal
Processing—Ft. Worth
Bodycote Thermal
Processing—Tulsa
Bomak Corp.
Bowdill Co.
Braddock Metallurgical
Braddock Metallurgical
—Alabama
Brite Brazing
Brite Metal Treating Inc.
Burbank Steel Treating Inc.
Cal-Doran Division
Carolina Commercial Heat
Treating
Caterpillar Industrial
Products Inc.
Central Kentucky
Processing, Inc.
Century Sun Metal Treating
Certified Heat Treating Inc.
Certified Metal Craft, Inc.
Cincinnati Flame
Hardening Co.
Cincinnati Gear
Cincinnati Steel Treating Co.
City Steel Treating Inc.
Clearing Niagara
Cleveland Flame Hardening
Coleman Commercial Heat
Treating
Commercial Induction
Commercial Steel Treating
Custom Heat Treating Co.
Delavan Steel Treating
Detroit Flame Hardening
Detroit Steel Treating Co.
Diamond Heat Treating Co.
Disston Precision Inc.
Dixie Heat Treating Co.
Dixie Machine & Heat
Treating Inc.
Drever Heat Treating
Dynamic Metal Treating
East Carolina Metal
Treating Inc.

East-Lind Heat Treat
Eckel Heat Treat
Economy Flame
Hardening, Inc.
Edwards Heat Treating
Service
Elmira Heat Treating, Inc.
Engineered Heat Treat, Inc.
Eric Steel Treating Inc.
Euclid Heat Treating
Fairfield Mfg. Co.
Feinblanking Ltd.
Fenton Heat Treating
Flame Metals Processing
Fluxtrol Manufacturing Inc.
Fox Steel Treating Co.
FPM Heat Treating
—Elk Grove
FPM Milwaukee
Franklin Steel Treating
Gear Company of America
General Heat Treating
General Metal Heat
Treating Inc.
Geo. H. Porter Steel
Treating Co.
Gibson Heat Treat Inc.
**Gleason Pfauter Hurth
Cutting Tools**
Grand Rapids Commercial
Heat Treating Co., Inc.
H & M Metal Processing
H & S Heat Treating
Hansen-Balk Steel Treating
Hauni Richmond Inc.
Heat Treat Corp. of
America
Heat Treating Services
Heat-Treating Inc.
Hinderliter Heat Treating Inc.
—Anaheim
Hinderliter Heat Treating Inc.
—Dallas
Hinderliter Heat Treating Inc.
—Tarzana
Hinderliter Heat Treating Inc.
—Tulsa
Hi-Tech Aero
Hi-Tech Metallurgical Co.
Hi-Tech Steel Treating Inc.
Hi-TecMetal Group
Corporate HQ
Horsburgh & Scott Co.
Horizon Steel Treating, Inc.
Houston Flame Hardening
HTG Aerobraz
HTG IMT-Duncan
HTG IMT-York
Hudapack Metal Treating
—Elkhorn
Hudapack Metal Treating
—Glendale Heights
Huron
Hydro-Vac
Hy-Vac Technologies Inc.
Illiana Heat Treating Inc.
Impact Strategies, Inc.
Induction Metal Treating
Induction Services Inc.
Inductoheat, Inc.
Industrial Metal Treating
Industrial Steel Treating Co.
Industrial Steel Treating Inc.
International Induction Inc.
Ipsen International Inc.
—Atmosphere Products
Ipsen International Inc.
—Vacuum Products
Ironbond Heat Treating
Irwin Automation Inc.
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East-Lind Heat Treat—Air, Oil, Water Hardening
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Fox Steel Treating—Select Hole Quenching
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H & S Heat Treating—Neutral Hardening
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Implantation Systems
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9th International Induction Heating Seminar

The Inductoheat Group is hosting its
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The world's largest group of experts from North America, Europe, Australia, South America and Asia will present information on NEW processes and developments in induction heating.

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- Billet/bar/wire heating
- Slab/strip/plate heating
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- Registration/reception Tuesday evening
- Formal presentations Wednesday & Thursday
- Roundtable discussions Friday until noon

It's Not Too Early to Register

Be sure to register via phone, fax, e-mail or our website to guarantee your spot in the seminar. The seminar fee is \$195.00 per person until April 10, 2000. Registration after the 10th is \$250.00 per person. Please make checks payable to Inductoheat, Inc.



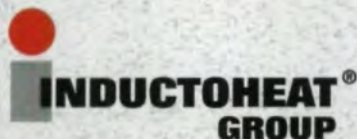
The 8th International Induction Heating Seminar, held in November, 1998, featured guest speakers and attendees from 19 countries.



The Inductoheat Group's 9th International Induction Heating Seminar will be held at the Hilton Clearwater Beach Resort in Clearwater, Florida.

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CIRCLE 161

Welcome to the Company Index of the 2000 Gear Technology Directory of Heat Treating Services. Use this index to locate the complete contact information for each company listed in the Services Index. Gear Technology advertisers are shown in boldface type. To find the pages on which their ads appear, see the Advertisers Index on page 17.

While we have made every effort to ensure that company names and addresses are correct, we cannot be held responsible for errors of fact or omission. If your company was not listed in this directory, and you would like to be included in the next one, please call 847-437-6604.

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March 29-31. Plastic Gears—Design and Manufacturing. University of Wisconsin, Milwaukee, WI. Offered through the Center for Continuing

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April 5-7. Fundamentals of Gear Design—Part 2 of 2. University of Wisconsin, Milwaukee, WI. Offered through the Center for Continuing Engineering Education, this is the second part of a 2-part course (Part 1 was held in December) and features Ray Drago as instructor. Topics include: manufacturing methods and considerations, inspection and quality control, materials and heat treatment, drawing data requirements and specifications, basics of load capacity rating and lubrication types and methods. For more information contact the registration office at (414) 227-3139 or log onto www.uwm.edu/dept/ccee.

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Gear Heat Treating by Induction

Dr. Valery Rudnev, Don Loveless, Brian Marshall,
Dr. Konstantin Shepeljakovskii, Norm Dyer and Micah Black

The induction hardening and tempering of gears and critical components is traditionally a hot subject in heat treating. In recent years, gear manufacturers have increased their knowledge in this technology for quality gears.

In contrast to carburizing and nitriding, induction hardening does not require you to heat a whole gear. With induction, the heating is localized to those areas where metallurgical changes are desired. The induction hardening process is a combination of electromagnetic, heat transfer and metallurgical phenomena that occurs when a workpiece (i.e. gear) is heated rapidly to a temperature above that which is required for a phase transformation to austenite and then rapidly quenched. One of the goals of induction hardening is to provide a fine grain martensitic layer on specific areas of the gear to increase hardness and wear resistance while allowing the remainder of the part to be unaffected by the process. Another goal deals with an ability to provide significant compressive stresses at the workpiece surface. This is a crucial feature, since it reduces crack propagation.

Induction heat treating is typically accomplished in a relatively short time and with high efficiency because energy is applied to the part only where it is needed. Induction equipment can be easily automated and



Fig. 1—Sample induction hardened gears.

incorporated into a work cell. The ability to heat treat in-line, as opposed to batch processing, provides high productivity and controllability and takes less shop floor space.

The kind of steel or iron used and its prior microstructure and gear performance characteristics dictate the required hardness profile, gear strength and residual stress distribution. Minimum gear shape distortion and pattern repeatability are among the most critical parameters that should be satisfied when heat treating gears.

Not all workpieces are well suited for induction heating. The best candidates are parts that have a classical geometry, including bushings, bars, pins, rings, plates, shafts, etc. External spur and helical gears, bevel and worm gears, internal gears, racks and sprockets are also among the parts that often undergo heat treating by induction (Fig. 1).

Hardening Patterns

The first step in designing an induction gear heat treatment machine is specifying the required hardness profile. There is a common misconception that a uniform contour pro-

file is always the best pattern for gear hardening applications. It is not. In many cases, a certain hardness gradient profile can provide a gear with better performance. Let's briefly evaluate a variety of hardening patterns (Fig. 2) and their effect on a gear's load carrying capacity and life.

Pattern A is a flank hardening pattern that has been used since the late 1940s for hardening large gears (outside diameter greater than 300 mm with tooth modules of 10–12 and larger). This pattern provides the required wear resistance, but the typical failure mode of gears with this type of pattern is a fatigue crack initiating at the tooth root area. It is typically strongly recommended that one use a pattern that hardens the root area as well, such as that pictured in pattern I.

Pattern B is a flank and tooth hardening pattern. This pattern has a similar shortcoming to the previous one, featuring poor load carrying capacity. It can be used in cases where wear resistance is of prime concern. However, patterns E, F and G provide better results

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is group director, science & technology, for Inductoheat, Inc. He has an M.S. in electrical engineering and a Ph.D. in induction heating. From 1993 to 1999, he worked as chief scientist for Inductoheat. He has 24 years of experience in induction heating. His credits include 14 patents and 92 scientific and engineering publications.

Don Loveless

is group vice president, research & development, for Inductoheat, Inc. He has 28 years of experience in induction heating. He concentrates on the research and development of high-/medium-frequency thyristorized and transistorized power supplies. Among his credits are 4 patents and 18 research/engineering publications devoted to different aspects of induction heating.

Brian Marshall,

applications engineer, joined Inductoheat, Inc. in 1979. He has been involved in the design and development of machines and processes for a variety of induction heat treating applications, including heat treatment of gears.

Dr. Konstantin Shepeljakovskii,

retired, has a Ph.D. in material science. Among his credits are more than 60 patents. He works as a consultant for Inductoheat, Inc. and NPO TechMash (Russia).

Norm Dyer,

international sales manager, joined Inductoheat, Inc. in 1989. He holds a B.Sc. degree from Wayne State University. Mr. Dyer has been involved in the improvement of a variety of induction metal heat treating processes.

Micah Black,

induction coil design supervisor, joined Inductoheat, Inc. in 1981. He has been involved in the development of a number of advanced induction processes and coil designs. Among his credits are 6 engineering/research publications.

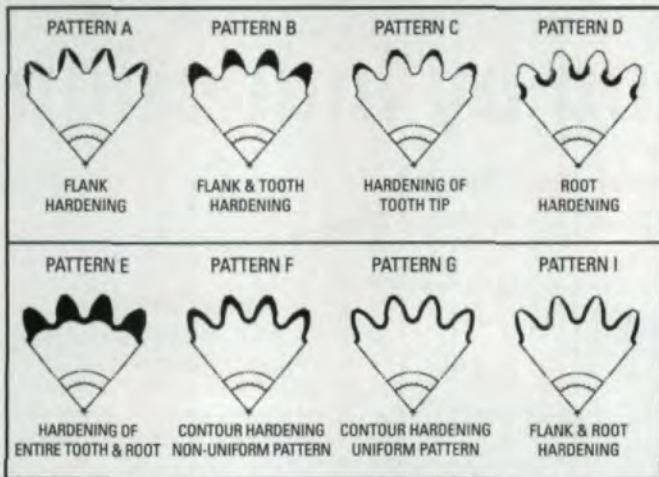


Fig. 2—Induction hardening patterns for gears.

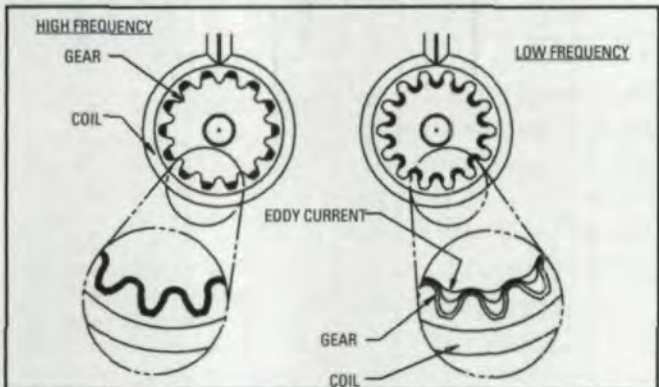


Fig. 3—Frequency influence on hardness profile with an encircling induction coil.

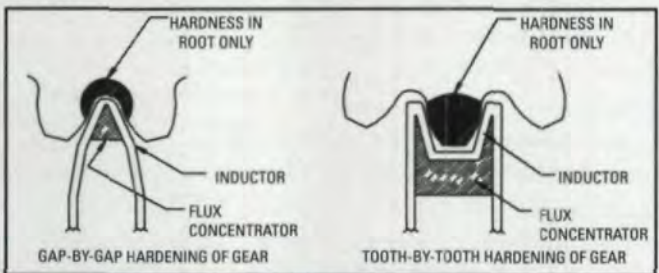


Fig. 4—Gap-by-gap and tooth-by-tooth induction hardening.



Fig. 5—Effects of changes in time, frequency and power on the hardening patterns in a steel shaft.

when wear, tear and fatigue resistance are required.

Pattern C is a tooth tip hardening pattern. In this case, the gear has minimum shape distortion. The application of gears with this pattern is extremely limited because the two most important gear areas (flank and root) are not hardened. In most cases, patterns F and G would be better choices.

Pattern D is a root hardening pattern. Application of this pattern is very limited as well, since it has poor wear resistance. Theoretically, it is possible to imagine the necessity of using this pattern as well as the previous one; however, practically, it is better to use another pattern, such as pattern I.

Pattern E is one of the most popular induction hardening patterns, particularly for small gears and sprockets. Since the body of the tooth is through hardened, there is a danger of brittle fracture in gears subjected to shock loads. Therefore, one typically applies a low-temperature tempering that lowers the final hardness down to 52–58 HRC. This pattern offers good resistance to wear and pitting.

Patterns F & G are popular patterns for medium size gears in many applications. Case depth at the root area is typically 30–40% of the depth in the tooth tip. It is very important to harden an entire gear perimeter, including flank and root area. A relatively ductile tooth core (28–44 HRC) and a hard surface (56–62 HRC) provide a good combination of such important gear properties as wear strength, toughness and bending fatigue.

Pattern I is one of the most popular choices for induction

hardening large gears and pinions (300 mm or more in outside diameter) with coarse teeth (modules greater than 10–12). This pattern provides an exceptional combination of fatigue and tear strength and shock resistance, which is very important for heavily loaded gears and pinions experiencing severe shock loads.

Coil Geometry and Heat Mode

The variety of required hardness profiles calls for different coil designs and heat modes. Development, including coil design, is largely based on induction principles, the results of mathematical evaluation and experience with previous jobs. The development establishes not only process parameters, including cycle times and power levels, but also coil geometry.

Tooth-by-tooth and gap-by-gap inductors. Generally speaking, gears are induction heat treated by either encircling the part with a coil (Fig. 3) or, in larger gears and pinions, heating them tooth-by-tooth or gap-by-gap (Fig. 4). Both tooth-by-tooth and gap-by-gap techniques can be realized by applying a single-shot or scanning mode. A gap-by-gap inductor can be designed to heat only the root and/or flank of the tooth, leaving the tip and the core soft and ductile. There are many variations of coil designs applying these principles. Probably one of the most popular is a “zigzag” shaped inductor.

Generally speaking, power requirements of both tooth-by-tooth or gap-by-gap hardening are relatively low, and applied frequencies are usually in the range of 1-10 kHz. At the same time, this is a time-consuming

process with a low production rate. Pattern uniformity is very sensitive to coil positioning. In addition, there is typically an appreciable shape or size distortion. Shape distortion is particularly noticeable in the last heating position. The last tooth can be pushed out by 0.1-0.3 mm. Therefore, final grinding is often required. Distortion can be minimized by hardening every 2nd tooth or tooth gap, (but this requires 2 revolutions to harden the entire gear). It is necessary to mention here that due to small coil-workpiece air gaps (0.5-1.5 mm) and harsh working conditions, the induction coils often require intensive maintenance and have relatively short lives compared with inductors that encircle the gear. When designing this type of inductor, particular attention should be paid to electromagnetic end/edge effects and the ability to provide the required pattern in the gear face areas.

Encircle inductors. When applying encircle coils, there are five parameters that play a dominant role in obtaining the required hardening pattern: frequency, power, cycle time, coil geometry and quenching conditions. Proper control of these parameters can result in totally different hardened profiles. Figure 5 illustrates a diversity of induction hardening patterns that were obtained on the same carbon steel shaft thanks to variations in time, frequency and power. As a basic rule, when it is necessary to harden the tooth tips only, a higher frequency and high power density should be applied (Fig. 3, left picture). When hardening the tooth root, a lower frequency and lower power density should be used (Fig. 3, right picture). A high

power density generally gives a shallow pattern; conversely, a low power density will produce a deep pattern.

Figure 6 shows three of the most popular design concepts of the induction gear heat treating processes that employ encircle-type coils: conventional single frequency concept (CSFC), pulsing single frequency concept (PSFC) and pulsing dual frequency concept (PDFC). All three concepts can be used in either a single-shot or scanning mode.

The conventional single frequency concept is typically used for hardening gears with small teeth. As one can see in Figure 2 (patterns B & E), the teeth are usually through hardened. Quite often, CSFC can also be successfully used for medium size gears. As an example, Figure 7 shows the induction gear hardening machine that applies this concept. The part being heat treated in this application is an automotive transmission component with helical teeth on the inside diameter and large teeth on the outside diameter. Both the inside diameter and the outside diameter require hardening (Fig. 8). The hardening of the inside diameter gear teeth requires a higher frequency than the outside diameter. Therefore, a frequency of 10 kHz was chosen for O.D. hardening, and a 200 kHz frequency was chosen for I.D. heating. Precise control of the hardening operations and a sophisticated design concept minimize part distortion and provide desirable residual stresses in the finished gear.

Gears are conveyed to the machine, where they are transferred by a cam-operated robot to the spindle of a heat treating



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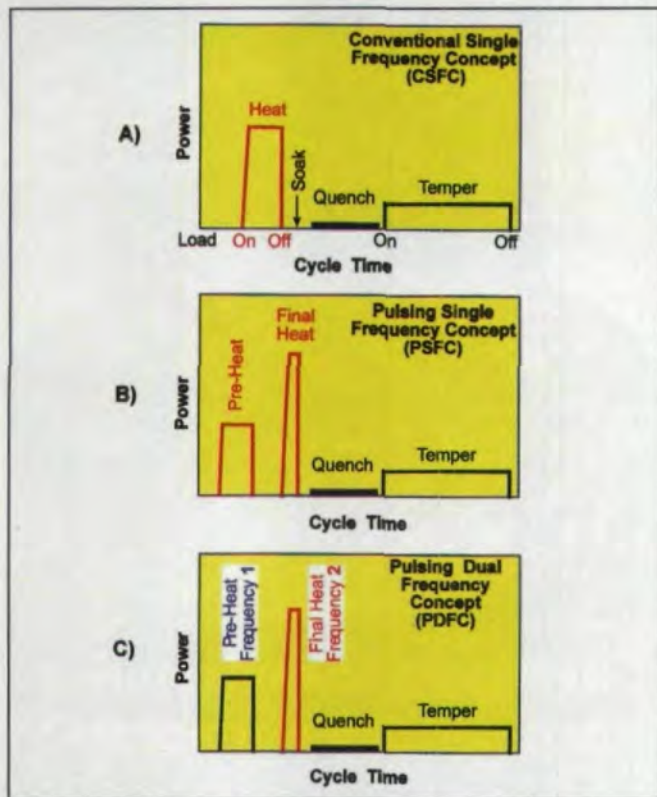


Fig. 6—Concepts of gear hardening by induction.



Fig. 7—Induction heat treating an automotive transmission gear with internal and external teeth.



Fig. 8—Cross section of an automotive transmission component after induction hardening.

station. Parts are monitored at each station and accepted or rejected based on all the major factors that affect gear quality. This includes energy input into the part; quench flow rate; temperature and pressure; and heat time. An advanced control/monitoring system verifies all machine settings to provide confidence in the quality of processing for each individual gear.

Quite often, in order to prevent problems such as pitting, spalling, tooth fatigue and endurance, it is necessary to harden a contour of the gear (contour hardening). In some cases, this can be a difficult task due to the difference in current density (heat source) distribution and heat transfer conditions within a gear tooth. Two main factors complicate the task of obtaining a required contour hardness profile.

The first factor is that with encircle-type coils, the root area does not have a good coupling with the inductor compared to the coupling at the gear tip. Therefore, it is more difficult to induce energy in the gear root. Secondly, there is a significant heat sink located under the gear root (below the base circle, Fig. 3). In order to overcome these difficulties and be able to meet customer specifications, the pulsing single frequency concept (PSFC) has been developed (Fig. 6b). In many cases, PSFC allows the user to avoid the shortcomings of CSFC and obtain a contour hardening profile. Pulsing provides desirable heat flow towards the root of the gear tooth without noticeable overheating of the tooth tip.

A typical "dual pulse" contour hardening system, which

applies a pulsing single frequency concept, has been discussed in Reference 2. This machine is designed to provide gear contour heat treatment (including pre-heating, final heating, quenching and tempering) with the same coil using one high frequency power supply. Figure 6b illustrates the process cycle with moderate power preheat, soaking stage, short high power final heat and quench followed by low power heat for temper. Preheating ensures a reasonable heated depth at the roots of the gear, enabling the attainment of the desired metallurgical result and decreasing the distortion in some materials. Obviously, preheating reduces the amount of energy required in the final heat.

A third concept—the pulsing dual frequency concept (PDFC)—is not a new one. The idea of using two different frequencies has been around since the late 1950s. This concept was primarily developed to obtain the contour hardening profile of helical and straight spur gears. Since several different companies, including Contour Hardening, Inductoheat and others, have pursued this idea, several different names and abbreviations have been used to describe it. However, regardless of the differences in nomenclature and the slight process variations, the basic idea is the same.

According to PDFC (Fig. 6c), the gear is preheated within an induction coil to a temperature determined by the process features that is usually 50–100°C below the critical temperature A_{c1} . Typically, this is accomplished by using a medium frequency (3–10 kHz).

Depending on the type of gear, its size and material, a high frequency (30–450 kHz) and high power density are applied during the final heat stage. For the final heating stage, the frequency selected allows the current to penetrate only to an exact repeatable depth. Quenching is done to complete hardening and bring the gear to ambient temperature. In some cases, dual frequency machines produce parts with lower distortion and more favorable distribution of residual stresses compared to other techniques.

The main drawback of this process is its complexity and high cost, since it is necessary to have two different power supplies. In some cases, it is possible to use one dual-frequency power supply instead of two single frequency inverters. However, the cost of these variable frequency devices is high, and their reliability is quite low.

Special attention should be paid when designing induction hardening machines for powdered steel gears. These gears are affected to a much larger extent by variations in the material properties of powder metals as compared to gears made by casting or forming. This is because the electrical resistivity, thermal conductivity and magnetic permeability strongly depend on the density of the powder metal.

TSH Technology for Gears

An impressive result can be achieved not only by developing a sophisticated process, but also by using existing processes with a combination of advanced steels. Through and surface hardening technology (TSH) is a synergistic combination of advanced steels and special induction hardening

techniques. These steels were invented by Dr. K. Shepeljakovskii (Ref. 6). The new low-alloyed carbon steels are characterized by very little grain growth during heating into the hardening temperature range. They can be substituted for more expensive, standard steels that are typically hardened by conventional induction, carburizing or quenching and tempering.

Main features of TSH technology include:

- TSH steels are relatively inexpensive, incorporating significantly smaller amounts (3–8 times less) of alloying elements such as manganese, molybdenum, chromium and/or nickel.
- Lower induction hardening frequency (1–10 kHz) reduces power supply cost.
- High surface compressive residual stresses (500 Mpa/73 ksi +).
- Hardened depth is primarily controlled by the steel's chemical composition and initial microstructure. This makes the heat treating process repeatable and robust.
- Reduced chance of overheating part edges and sharp corners due to end effect.

Figure 9 shows an induction heat treated gear made from TSH steel. One of the unique features of that gear is that instead of using a two-step approach (first O.D. heat and then I.D. heat, or vice-versa), that gear has been heated and quenched in a single step using only one inductor. O.D. and I.D. teeth have fine grained martensite with a hardness of 62 HRC. The microstructure of the core is a combination of very fine pearlite and bainite having a hardness of 25–40 HRC.

TSH technology parts are stronger and more durable than

some made of conventionally heat treated standard steels. Typical applications include gears, bushings, shafts, coil springs and bearings (Ref. 6).

Induction Tempering

The stress relieving/tempering process takes place after the part is hardened. It is a subsequent but no less important step in metal heat treating. The main purpose of tempering is to decrease the gear brittleness without causing too great a decrease in the as-quenched hardness, to relieve internal stresses, and in some cases to improve shape stability (Ref. 5).

A conventional method of tempering induction hardened gears is to heat them in an oven or a gas-fired or infrared furnace, which is typically located in another area of the plant. This has penalties in terms of floor space, labor and time needed to transport parts. In addition, a furnace tempering operation may take two to three hours to complete. Short-time induction tempering was developed to overcome these drawbacks.

Time and temperature are two of the most critical parameters in short-time induction tempering. However, temperatures higher than those used for furnace tempering must be used to provide a similar effect. There are several ways to determine the time-tempera-

ture correlation between conventional long-time, lower temperature furnace tempering and short-time, higher temperature induction tempering, including, for example, the Hollomon-Jaffe equation and the Grange-Baughman tempering correlation.

There is a common misconception that tempering removes all internal stresses. Tempering does decrease some stresses. It makes the steel softer and reduces the chance of noticeable distortion and the possibility of cracking. As a matter of fact, it is not really desirable to relieve all stresses. As mentioned above, in most gear heat treating applications, the existence of good compressive residual stresses at the gear surface is useful and very desirable since it reduces the possibility of crack development.

There is a balance of residual stresses in the workpiece. Therefore, if in certain areas of the gear there are compressive residual stresses, then somewhere within the workpiece, there must be tensile stresses. Applied stresses are maximum at the gear surface and then rapidly drop off. Therefore, one of the important "duties" of tempering is not only the reduction of tensile stresses, but also the shifting of the maximum of these stresses toward the core.



Fig. 9—An induction heat treated gear made from specialty TSH steel.

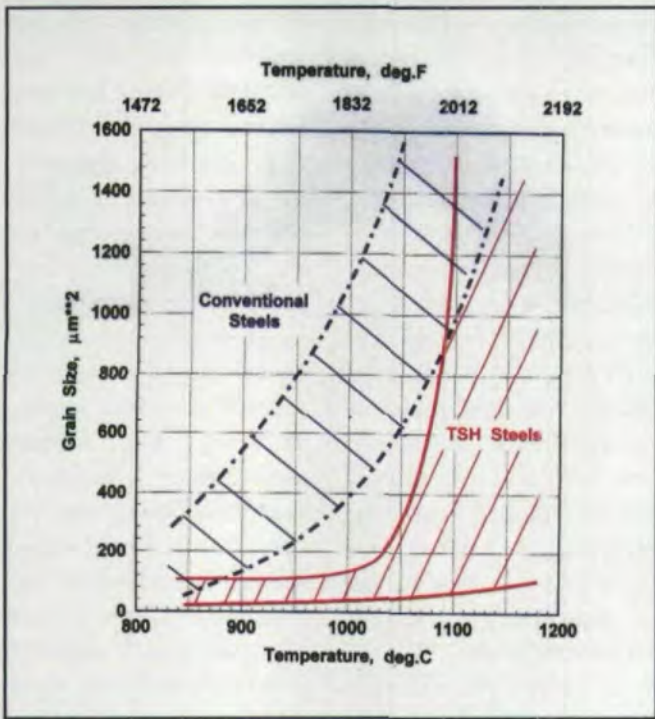


Fig. 10—Grain growth for TSH steels vs. that of conventional grades.

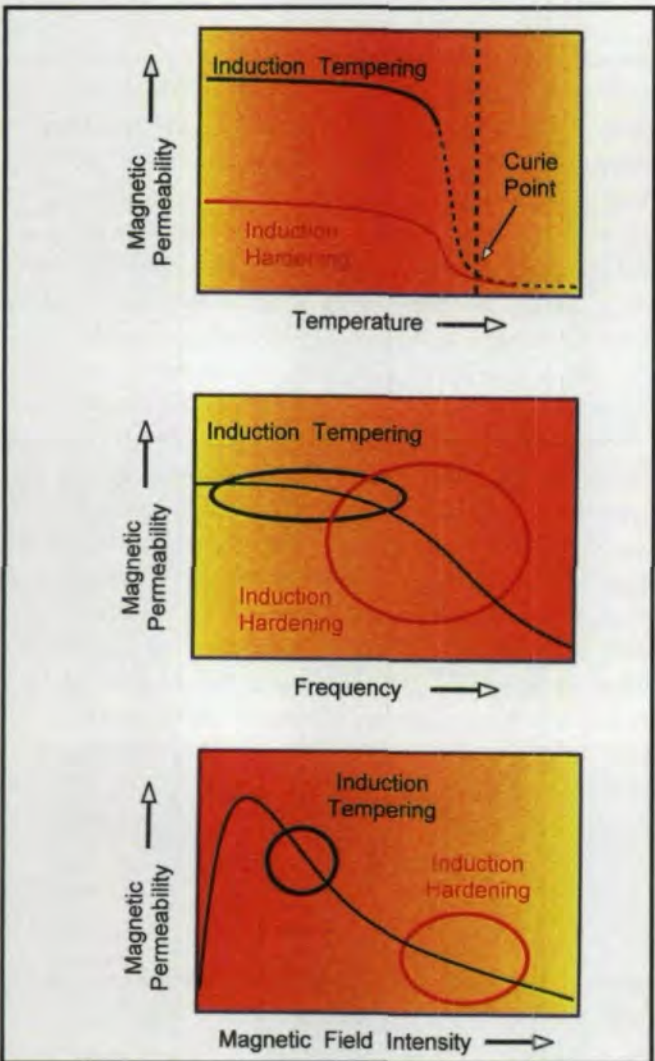


Fig. 11—Magnetic permeability vs. temperature, frequency and magnetic field intensity.

It is important that the time from quench to temper be held to a minimum. If this "transient time" is long enough, the internal stresses may cause a noticeable size and shape distortion, or even cracking. Therefore, a long transient time between quenching and tempering will decrease or eliminate the tempering benefits.

In the case of induction tempering any complex parts (including gears), the choice of frequency, power density and coil geometry is dictated by the need to apply enough energy into certain areas of the part. In gear tempering applications, it is necessary to induce enough energy into the root area of the tooth without overheating its tip.

The root of the gear is a critical area because the maximum concentration of stresses is typically located there. As a result, fatigue cracks occur primarily in the root area. Therefore, it is very important for this area to be stress relieved. There are three factors that make this task quite a complicated one. Two factors are similar to hardening and were discussed above. One of them deals with poor electromagnetic coupling between the coil and the tooth root compared with the tooth tip. Another one deals with the existence of a heat sink phenomenon in the root. The third factor derives from the fact that the tempering temperatures are always below the Curie point. Therefore, the gear is magnetic and the skin effect is always pronounced (Fig. 3, left figure). The use of high frequency for induction tempering will result in an essential power surplus in the tip of the tooth com-

pared to its root and will have a tendency to overheat edges and sharp corners. In order to overcome these difficulties, low frequency, loose coil coupling and low power density should be used for tempering.

As discussed above, it is possible to harden and temper gears in the same coil using the same power supply. In some cases, it is the best concept and has an obvious low capital cost advantage and less tooling to store. In other cases, it might not best suit customer requirements.

Since the power density required for tempering is quite low, it is necessary to heat a gear at a slow rate to avoid tooth tip overheating. Depending upon the type of power supply, this is not always an easy task from the load matching point of view. In addition, the depth of current penetration in carbon steel at tempering temperatures is very small compared to its value during hardening. This is due to the fact that tempering temperatures are always below the Curie point, and therefore, steels are always in a magnetic state. In addition, the relative magnetic permeability of steel during induction tempering is more than ten times higher compared to the permeability of steel during induction hardening. This is due to the low magnetic field intensities used in induction tempering (Fig. 11).

A substantial increase in magnetic permeability results in a significant decrease in penetration depth of an induced current. Therefore, in order to heat a gear for tempering to the same depth as hardening, it is wise to use a lower frequency.

In addition, time required for induction tempering is typically 2-4 times that of induc-

tion hardening. Therefore, when one uses the same coil for hardening and tempering, the production and power supply utilization might suffer.

Heating for tempering with a separate coil and dedicated power supply is a more costly solution from a capital investment point of view, but at the same time, it has several noticeable advantages. The current distribution and power density can be optimized specifically for the tempering operation. A separate, loosely coupled, channel-type, single- or multi-turn coil can be used effectively for this purpose. Equipment will be used very effectively with high production. One hardening machine can operate in conjunction with two or three tempering machines.

The decision to induction temper should be carefully weighed (Ref. 5). Some metallurgists are not comfortable with tempering for a short time and then only in the hardened area. They feel that furnace heating of the entire part and holding it a temperature for hours, vs. seconds or minutes, is more reliable. The key to any gear or critical component production process is how well the finished part performs in service. An induction tempered gear, like any other machine component, should be thoroughly tested and evaluated for reliability. Necessary test data for induction and furnace tempered parts should be compared. It is important to remember that the surface temperature alone is not a valid indication of a proper temper. If tempering has been done correctly, there will be only a slight reduction in hardness, which will be more than offset by the benefits obtained,

including internal stress relief, improved ductility or toughness, and shifting of the maximum tensile stress farther away from the applied stress.

The advantages of induction tempering—system compactness, single part processing, energy efficiency, and precise control and monitoring of an individual part—in many cases far outweighs the disadvantages and fear of the untried.

Conclusion

Space limits this discussion to major features of induction gear heat treating. There are many aspects involved in designing and manufacturing contemporary gear hardening and tempering systems. This includes an effect of prior microstructure and grain size on hardening pattern. Others deal with quenching, cracking, shape/size distortion and residual stress distribution. ⚙

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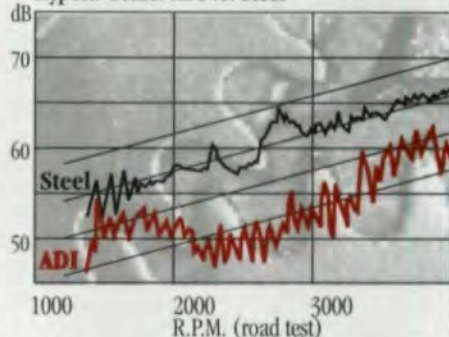
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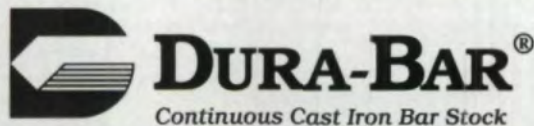
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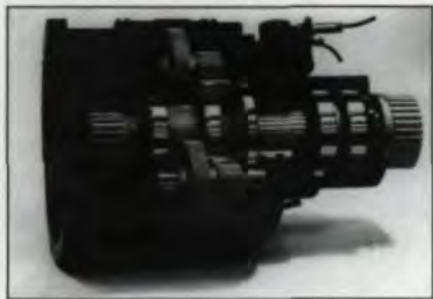
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New 2-Speed Machine Tool Gearbox

"Built-in" RAM-MSD 2-speed gearboxes, available from Andantex USA, Inc., are integrated in-line between the water-cooled motor and the spindle inside of the machine tool's RAM. These units extend the constant power speed range of the motor, providing high output speeds (8,000 rpm) for finishing aluminum and high torque at low speeds for hogging out steel or cast iron. Standard ratios are 1:1, and reductions range from 2.5:1 to 5:1.

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Two New Items from Ogasawara

The GRT-04 Gear Rolling Tester is designed to measure the composite error of fine pitch gears relative to the function of center distance oscillation during dual flank meshing between the test piece and the master gear. This machine is suitable for either the lab or the shop floor. The unit can also be used to measure other gear products such as plastic gears and motor shaft gears up to a foot in length. Center distance deviations are detected continuously with the GRT-04's small

displacement detecting system, which utilizes a differential capacitance device for frequency modulated (FM) signal output. With optional data analyzing software, you can classify your gear products to meet with a specified gear accuracy such as AGMA, ISO or DIN.

The new HBS-1520 Hob Cutter Sharpening Machine is the result of a collaboration between Ogasawara, which designed the machine, and Saikuni, which built it. Using Ogasawara's optimal dressing method, the HBS-1520 can achieve both a mirror finish on the cutting surface and extreme geometrical accuracy. Carbide hobs can be sharpened using a diamond wheel and helical flutes can be ground with the optional spindle head. The machine is easy to set up and operate and it can reduce grinding costs with its automatic grinding procedures.

For more information, contact Ogasawara Precision Hob Lab, Ltd. at +(81) 44-877-3511 or visit their Web site at www.og-pl.co.jp. You may also contact their American representatives, Russell, Holbrook and Henderson, Inc., at (201) 670-4220 or visit their Web site at www.tru-volute.com.

Circle 301



New RV Gear Reducer from Harmonic Drive

Harmonic Drive Technologies introduces the Rotary Vector (RV)-C hollow shaft gear reducer. The RV-C gear is open through the center, allowing vacuum lines, wiring harnesses, concentric shafting and cooling lines to be run through it. Hole

sizes range from 31 mm (1.2") to 138 mm (5.4"). The RV-C also has specially designed, built-in output bearings that support large thrust and overhung loads.

The RV-C is one of three heavy-duty RV power transmission configurations manufactured by Teijin Seiki and supplied by Harmonic Drive Technologies. These are precision drives that offer high-ratio gear reduction in a compact design. RV drives offer very high torsional rigidity and an overload torque capacity of 500% of the unit's rated torque. Total lost motion from all sources including backlash, spring rate and hysteresis, is limited to one arc-minute. For more information call (800) 921-3332 or visit Harmonic Drive Technologies' Web site at www.harmonic-drive.com.

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Rotary Diamond Dressing Roll Holds Tip Longer

Norton Company introduces a line of patent-pending precision rotary diamond roll dressers that maintain tip geometry longer, when CNC-profile-dressing intricate forms into conventional and superabrasive wheels.

The new Lap-Free BPR Profile Dressing Rolls are different from conventionally manufactured infiltrated and plated profiling rolls, which often use polycrystalline diamond (PCD) or natural diamond material. They are designed not to require relapping and are available in forms with included angles not previously available for CNC profile dressing.



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PRODUCT NEWS

With the elimination of relapping, the previously difficult task of estimating roll life now requires only measuring the roll diameter while it is mounted on the machine. Field tests have shown that the Lap-Free BPR Roll not only holds its tip radius longer, but lasts nearly three times longer than a conventional PCD roll in production dressing applications. For more information contact Sue Auburg at Norton Company, (800) 446-1119 or visit their Web site at www.nortonabrasives.com.

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Carpenter Project 7000® 15Cr-5Ni stainless steel (UNS S15500) has been developed by Carpenter Technology Corp., Reading, PA, to provide superior machinability while meeting all the requirements of Aerospace Material Specification AMS 5659 covering bars, wire, forgings, rings and extrusions. The new alloy, offering potential for increased capacity and longer tool life, is available as a "drop-in" replacement for conventional 15Cr-5Ni stainless in applications where improved machining productivity is desired.

Possessing high strength and hardness, along with excellent corrosion resistance, the new precipitation hardening stainless alloy is a candidate material for rod-end bearings, a variety of aircraft structural components and some engine parts. The new alloy may be considered a more machinable version of the standard 15Cr-5Ni stainless, which has been used for industrial applications such as gun barrels, valve parts, fittings and fasteners, shafts, gears and process equipment. For additional information or a technical data sheet, contact Jim Dahl by phone at (610) 208-2235 or by e-mail at jdahl@cartech.com.

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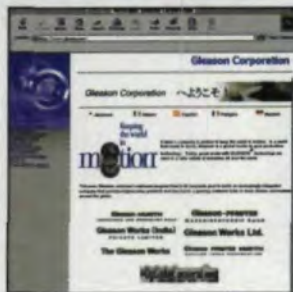
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MARKING TIME WITH WOOD

Gear Technology's bimonthly aberration — gear trivia, humor, weirdness and oddments for the edification and amusement of our readers. Contributions are welcome.

Clocks with wooden gears? In these days of gears made from plastic, steel and exotic materials; it is a little unusual to hear about a practical application for wooden gears. But that is exactly what David Scholl, the owner of Changing Times, a Harlingen, TX, clockmaker, is offering us.

His company specializes in making wooden geared clocks. It is a one-time hobby that Scholl made into a business. "I made my first wooden clock in about 1980," says Scholl. "At the time, I was attending college and working on an Industrial Arts/Business Administration degree. While thumbing through a woodworking magazine, I found an article describing the basic principles involved in clock mechanisms. Eventually, I found the name and address of a company under the direction of R. D. Thomas. Mr. Thomas sold (sells) plans for wooden



A wooden geared clock. Courtesy of Changing Times.

geared clocks, so I ordered them and got to work." The clocks he makes today are weight driven and can be made to run from 24 hours up to eight days depending on the drive train.

That drive train is made up of wooden spur gears with involute teeth. These are not little, fine pitched gears, either. "Because wood is less dense than other materials, there is an increased need for structural mass in order to add strength and stability to the gear teeth. Therefore, both for appearance and function, larger DP sizes are used beginning at 10 DP and going up," says Scholl, who adds that the gears are made using modified woodworking equipment. "Table saws have been modified to accept standard milling cutters," he says. "Many jigs and fixtures are needed in order to accommodate the many sizes of gears that we make. The rough cut gears are hand-faced on a wood lathe. Also, larger gears are laminated from up to 18 different pieces of wood and then cut. This is because wooden gears are more susceptible to dimensional changes due to temperature and humidity than are gears made from other materials. This means that warping is a factor to consider. The lamination process minimizes this factor."

The average sized gear has between 30 and 72 teeth. The diameter of a 30-tooth gear with a 7 DP is approximately 4.5 inches. A gear with 72 teeth would have a diameter of 10.5 inches. Pinions start at 8 teeth and are approximately 1.25 inches in diameter. And, according to Scholl, these gears will last. "Because of the low torque and low revolutions-per-minute, these gears should last quite



Wooden Gears. Courtesy of Changing Times. some time." He also adds that no grease or oil is used with the wooden gears. In fact, no lubrication is used on the teeth surfaces at all. "There are ball bearings mounted on the gear arbors in order to reduce start up friction," he says. "This is necessary because, in reality, the clocks stop and restart every second."

Changing Times sells the wooden-gear clocks through local specialty shops, but Scholl plans to open a Web site soon. Until then, if you want more information about Changing Times and its wooden-gear clocks, send e-mail to Scholl at chnetimes@aol.com. ☉

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